

N, S-Codoped TiO₂/Fe₂O₃ Heterostructure Assemblies for Electrochemical Degradation of Crystal Violet Dye

Pooja Sree, Palukuru; Vishnu Priya, Devangam A.; Dilip Kumar, Behara**

Department of Chemical Engineering, JNTUA College of Engineering, Ananthapuramu, INDIA

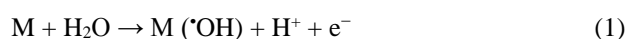
ABSTRACT: In contemporary research, “Heterostructure” assemblies play an important role in energy conversion systems, wherein the composite assemblies facilitate faster charge carrier transport across the material interfaces. The improved/enhanced efficiency metrics in these systems (electro/photo-electrochemical processes/devices) is due to synergistic interaction and synchronized charge transport across material interfaces. Herein, we report Type-I Heterostructure consists of N, S doped TiO₂ and Fe₂O₃ for electrochemical crystal violet dye degradation studies. Synthesized N-S codoped TiO₂/Fe₂O₃ composite heterostructured assemblies were fabricated on Titanium (Ti) substrate and used for electrochemical analysis. Complete decolorization was achieved with all the fabricated electrodes and a higher rate of degradation was achieved with the composite electrode (Ti/TiO₂/Fe₂O₃) in comparison to individual electrodes (bare Ti, Ti/TiO₂, Ti/Fe₂O₃). Further, a probabilistic mechanism of degradation is proposed in support of the hypothesis. The outcomes of present work will have a profound effect on doped semiconductor heterostructure assemblies in the degradation of complex dye molecules of industrial outlets.

KEYWORDS: Type-I Heterostructure, TiO₂, Fe₂O₃, Crystal violet dye.

INTRODUCTION

Electrochemical oxidation process emerged as an important technique for the degradation of toxic or biorefractory organic pollutants from waste water [1-2]. As electrons are the sole agents involved in the organic pollutant abatement, these methods are eco- friendly, environmentally compatible and mostly preferred [3-4]. Further, they possess specific advantages over traditional methods such as the elimination of by-products formation, no additional requirement of chemicals, easy operation, high efficiency, cost-effective and provides safety as it operates under mild conditions [5]. An electrochemical process can be either direct or indirect oxidation processes. In direct oxidation, direct

transfer of electron takes place at the anode and thereby degrade the pollutant adsorbed on the anode surface. In indirect oxidation, generation of hydroxyl radicals (*OH) takes place at the anode. The generated *OH radicals act as strong oxidizing agents and readily react with organic pollutants, thereby oxidizing the pollutants to small molecules such as CO₂ and H₂O [3-8]. The electrode reaction at the anode is



Where M represents anode. The electrochemical generation of *OH radicals and their chemical reactivity depends on the nature of the anode material used [9].

* To whom correspondence should be addressed.

+ E-mail: dileepbh.chemengg@jntua.ac.in

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Therefore, electro-catalytic materials which are used as anode plays a significant role in electrochemical oxidation processes [10]. Hence, the selection and design of anode/catalytic material are crucial and specific characteristics such as high stability, electrochemical/catalytic activity, over potential demands etc., has to be considered for effective degradation of organic pollutants [11-12].

In the recent past, semiconductors were used as noble anode materials in various catalytic/electro-catalytic/photo-catalytic/photo-electrochemical applications such as dye degradation, water splitting, CO₂ separation, and hydrogen storage [13-16]. It is mainly due to their extensive physical, chemical and readily tunable/modifiable properties either by size reduction or doping or sensitization to acquire desired properties of interest [17]. Semiconductors such as TiO₂ and Fe₂O₃ have attracted a great deal of attention for degradation of organic pollutants in comparison to other semiconductor materials owing to their hydrophilicity, abundant availability, stability, environmental compatibility, low cost and corrosion resistance [18-20]. However, the application of TiO₂ is limited due to its high band gap. Therefore, efforts have been made to decrease the optical band gap as well as to improve the electronic conductivity [21]. One way to achieve this is by doping with nonmetals such as nitrogen (N) and sulphur (S) which are *iso*-electronic with TiO₂. The added dopants will occupy the interstitial sites/oxygen vacancy sites of TiO₂ and lead to the formation of localized states just above the valence band [21-22]. Further, the formation of Type-I Heterostructure with a Fe₂O₃ material (having suitable band edge alignments with TiO₂) will facilitate charge carrier flow across material interfaces and improve the electrochemical activity [23]. Therefore, the Heterostructure assembly will promote charge carrier separation and increase the charge carrier lifetime owing to high adsorption capacity, and effective surface area [24].

Herein, we propose electrochemical degradation of crystal violet dye using N-S co-doped TiO₂/Fe₂O₃ composite heterostructured assemblies fabricated on titania (Ti) substrate. Titania substrate is chosen as good support due to its stability, and better conductivity [25-26]. It is hypothesized that the formed heterojunction will facilitate the faster charge carrier movements due to

proper band edge alignments (CB edge of TiO₂ is -4.21 eV and CB edge value of Fe₂O₃ is -4.78 eV w.r.t Absolute Vacuum Scale [27]) and hence there is a net movement of charge carriers from TiO₂ to Fe₂O₃, which breaks the dye molecule at a faster rate. The fabricated electrodes were characterized using X-Ray Diffraction (XRD), Scanning Electron Microscopy (SEM), Energy Dispersive X-ray (EDX) spectroscopy, and Fourier Transfer InfraRed (FT-IR) spectroscopy. Further, to monitor electrochemical activity, analytical techniques such as cyclic voltammetry and UV-Vis spectroscopy measurements are used. The faster rate of decolorization is achieved with N, S-TiO₂/Fe₂O₃ electrode (~9h) in comparison to individual electrodes. This signifies the importance of iso-electronic non-metal (N, S) doped metal oxide Heterostructure in facilitating the fast charge carrier movements and thereby enhancing the electrochemical performance. The outcomes of the current article help not only in understanding the mechanism of dye degradation of complex molecules but also help to design low-cost catalytic Heterostructure assemblies for different catalytic/electro-catalytic applications.

EXPERIMENTAL SECTION

Materials and methods

Titanium isopropoxide [Ti{OCH(CH₃)₂}₄], Iron(III) nitrate nonahydrate (Fe(NO₃)₃.9H₂O), Ethanol (C₂H₅OH), Ammonia(NH₃), Thio-urea ((NH₂)₂CS), Sodium sulphate (Na₂SO₄), and Isopropyl alcohol (C₃H₈O) were purchased from Sigma Aldrich Chemicals (India) Ltd. Nafion(10%) liquid solution was purchased from Dupont Chemicals, USA. Crystal violet dye purchased from Sigma-Aldrich is used as a model pollutant in the present study. All the reagents were analytically grade and used as received without further purification. All solutions were prepared with distilled water.

Synthesis of un-doped and doped TiO₂ nanoparticles

9 mL of TTIP (Titanium tetra isopropoxide) is added dropwise to the homogeneous mixture of 150 mL of ethanol and 3.72 mL of distilled water after stirring for half an hour. The resultant mixture is stirred for 4 h at 85°C under continuous magnetic stirring followed by centrifuge, dried at 60°C in a hot air oven for 30 min and calcined at 400°C for 3 h. Thus, Anatase TiO₂ nanoparticles are obtained. By adding the ammonia [5 mL], thiourea [9.3 g] as a precursor to the

above procedure N-TiO₂, S-TiO₂ nanoparticles are obtained respectively and adding both (2.5 mL, 5.4 g) N, S-TiO₂ nanoparticles are obtained [28]. The continuous magnetic stirring process was carried out in a fume hood.

Synthesis of Fe₂O₃ nanoparticles and N, S TiO₂/Fe₂O₃ nanocomposite

4 g of Fe (NO₃)₃ · 9H₂O was dissolved in 100 mL of distilled water and was stirred for 30 min. 1.5 g of gelatin was added to 100 mL of distilled water and stirred for 30 min at 60°C. Then the gelatin solution was slowly added to the iron nitrate solution and stirred for 1 h. Then the obtained gel is dried in hot air oven at 60°C for 6 h and calcined at 600°C for 1 h [29]. Thus Fe₂O₃ nanoparticles are formed. Adding 10 mg of prepared N, S-TiO₂ nanoparticles to the iron nitrate solution and following the above procedure then the N, S-TiO₂/Fe₂O₃ nanocomposite is obtained. Synthesis procedure of N, S-TiO₂/Fe₂O₃ nano-composites is shown in Fig.1.

Electrode fabrication

40 mg of nanoparticles/composite particles (i.e. TiO₂, Fe₂O₃, N, S-TiO₂/Fe₂O₃) was mixed with 2 mL of isopropanol and 20 μL of Nafion solution. Then, it was sonicated for 1 h and the slurry was applied onto the Ti plate (effective electrode surface area~ 1cm²), and dried at 60°C on hot plate.

RESULTS AND DISCUSSION

Characterization of nanoparticles

XRD analysis

To study the crystalline nature of samples, XRD spectra of synthesized samples were taken from X-ray diffractometer with Cu K_α radiation ($\lambda = 1.5406 \text{ \AA}$). Fig. 2 represents XRD spectra of TiO₂, N-TiO₂, S-TiO₂, N, S-TiO₂, Fe₂O₃ and N, S-TiO₂/Fe₂O₃ nanoparticles. The intense peaks at 2θ values of TiO₂, N-TiO₂, S-TiO₂, N,S-TiO₂ were observed at 25.48°, 37.8°, 47.8°, 53.8°, 62.6° which corresponds to (101), (004), (200), (105), (024) atomic planes respectively confirms the formation of tetragonal Anatase phase matching with JCPDS 21-1272 (Fig. 2(a)-2(d)) [21, 30-31]. The average crystallite size estimated for TiO₂, N-TiO₂, S-TiO₂, N, S-TiO₂ will vary in between 4-7 nm from Debye-Scherrer formula. The diffraction peaks of N-TiO₂ and S-TiO₂ samples were not shifted in comparison to TiO₂ sample indicating there is no crystal distortion upon doping. Also, the increase

in intensity and narrowing the width of the diffraction peak of Anatase phase was observed for doped TiO₂ (Fig. 2(b)-2(d)) compared to pure TiO₂ (Fig. 2(a)). The results show that the doping of N and S into TiO₂ will increase the crystallite size and crystalline nature of the sample significantly.

Similarly, Fe₂O₃ exhibits peaks at 2θ values of 24.44°, 30.2°, 35.76°, 43.4°, and 62.91° corresponds to (012), (104), (110), (202), (214) indices confirms α- phase of Fe₂O₃ with JCPDS 03-0800 from Fig. 2(e) [32]. The average crystallite size of Fe₂O₃ nanoparticles was estimated to be 13 nm. Further, the diffraction peaks of N, S-TiO₂/Fe₂O₃ composite confirms the presence of both TiO₂ and Fe₂O₃ peaks as evident from Fig. 2(f). Moreover, the intensity of Anatase and α- phase peaks of N, S-TiO₂/Fe₂O₃ are attenuated signifying that they hinder the crystal growth of TiO₂ or Fe₂O₃ particles. Overall diffraction patterns from XRD spectra indicate the highly crystalline nature of Anatase phase TiO₂, α-Fe₂O₃ and N, S-TiO₂/Fe₂O₃ samples.

SEM and TEM analysis

The surface morphology of nanoparticles can be determined using Field Emission Scanning Electron Microscopy (FESEM), and elemental composition can be obtained by Energy Dispersive X-ray Spectroscopy (EDS). The surface material distribution of TiO₂ and N-S TiO₂ is shown in Fig. 3. However, the distribution is not uniform after doping in case of N, S-TiO₂. It may be probably due to surface segregation of the added dopants, and the lattice was completely distorted which was confirmed from SEM images Fig. 3. The spherical shape of TiO₂ was observed from Fig. 3(a). Inset Fig. 3(a) represents the EDS spectra which confirms the presence of Ti and O in the synthesized nanoparticles (Ti - 50.68%, O - 49.32%). The irregular shape of N, S-TiO₂ was observed from Fig. 3(b) which may be due to slight destruction in crystal lattice upon doping. Further, the surface texture and rough morphology shown in co-doped sample in comparison to pristine TiO₂ confirm the change in an electrochemical surface area available for catalytic activity. Further, to confirm the heterostructure formation, TEM analysis is performed (Fig. 3c). It can be clearly observed that the nanostructures are composed of N-S, co-doped TiO₂ and Fe₂O₃ which are widely distributed on entire surface as heterostructure

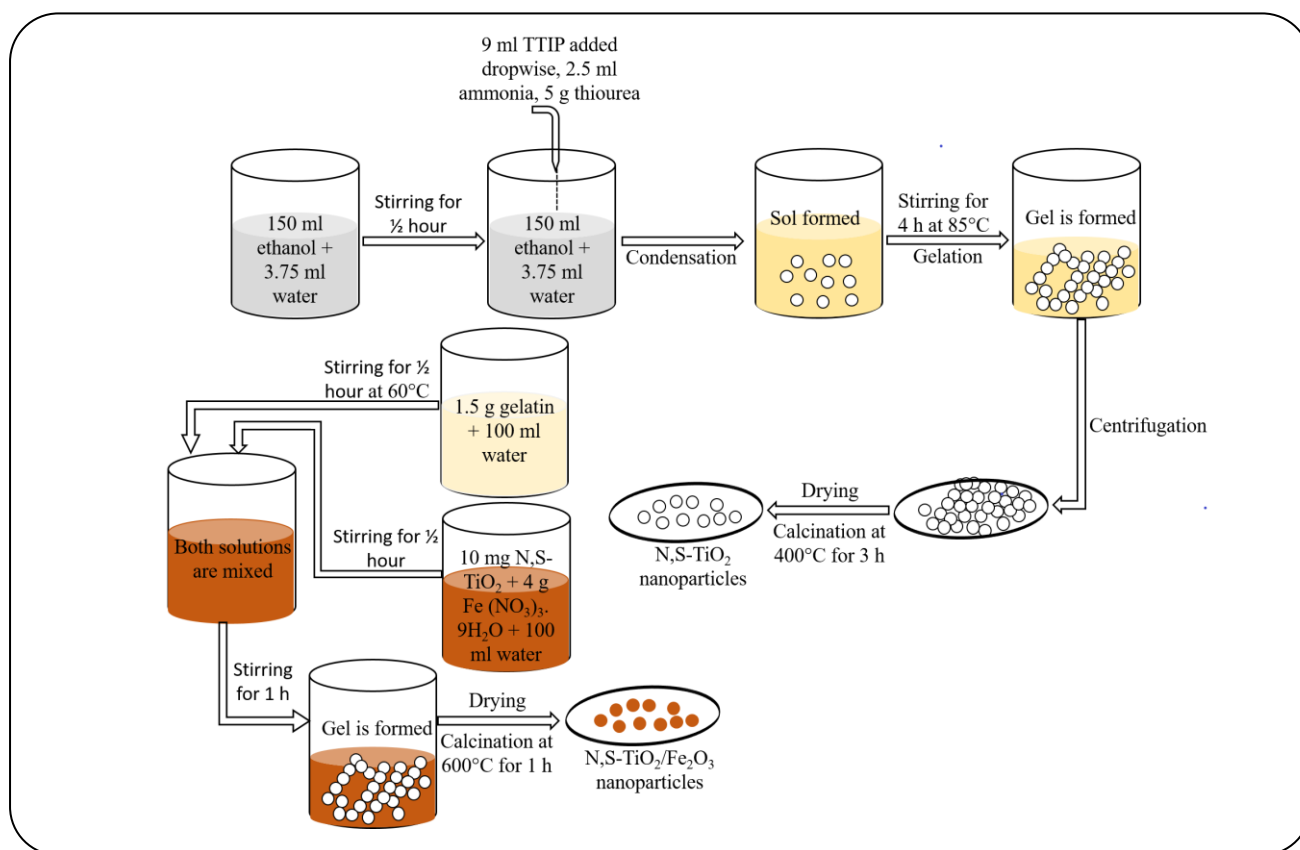


Fig. 1: Experimental procedure for the synthesis of $N, S\text{-TiO}_2/\text{Fe}_2\text{O}_3$ nanoparticles.

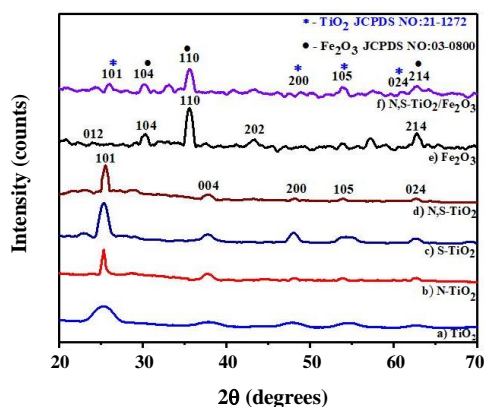


Fig. 2: XRD pattern of (a) TiO_2 , (b) $N\text{-TiO}_2$ (c) $S\text{-TiO}_2$ (d) $N, S\text{-TiO}_2$ (e) Fe_2O_3 (f) $N, S\text{-TiO}_2\text{-Fe}_2\text{O}_3$ nanoparticles.

assembly/composite matrix (Fig. 3c, 3d). The d-spacing observed to be 0.24 nm corresponds to (103) plane of TiO_2 and 0.37 nm corresponds to (012) plane of $\alpha\text{-Fe}_2\text{O}_3$ (Fig. 3d) which confirms the presence of two materials in heterostructure formation (the inset image shows zoomed portion of lattice fringes of Fe_2O_3).

FT-IR analysis

The FT-IR spectrum gives the information about the surface functional groups present on the surface of nanoparticles. The broadband observed at 3483 cm^{-1} represents the presence of hydroxyl groups on the surface of the sample which is significant factor in affecting electrochemical activity. The more the number of hydroxyl groups the faster the electro-catalytic degradation. The band at 2020 cm^{-1} signifies the deformation vibration of OH groups linked with titanium atoms (Ti-OH), and the band at 665 cm^{-1} corresponds to titania crystal lattice vibrations from Fig. 4(a)-4(d). The band at 3418 cm^{-1} and 1854 cm^{-1} were assigned to N-H stretching and bending vibrations of ammonium species from Fig. 4(b). No peak at 3483 cm^{-1} (OH) indicates that nitrogen is present only in the form of ammonium group on the surface. The band at 2782 cm^{-1} implies the -NCS stretching vibrations indicates the presence of nitrogen and Sulphur linkages on the surface of $N, S\text{-TiO}_2$ sample, and the surface adsorbed SO_4^{2-} species has a band at 1256 cm^{-1} corresponds to the asymmetric stretching vibrations of S=O from Fig. 4(c),4(d) [33-34]. The peak at 583.54 cm^{-1}

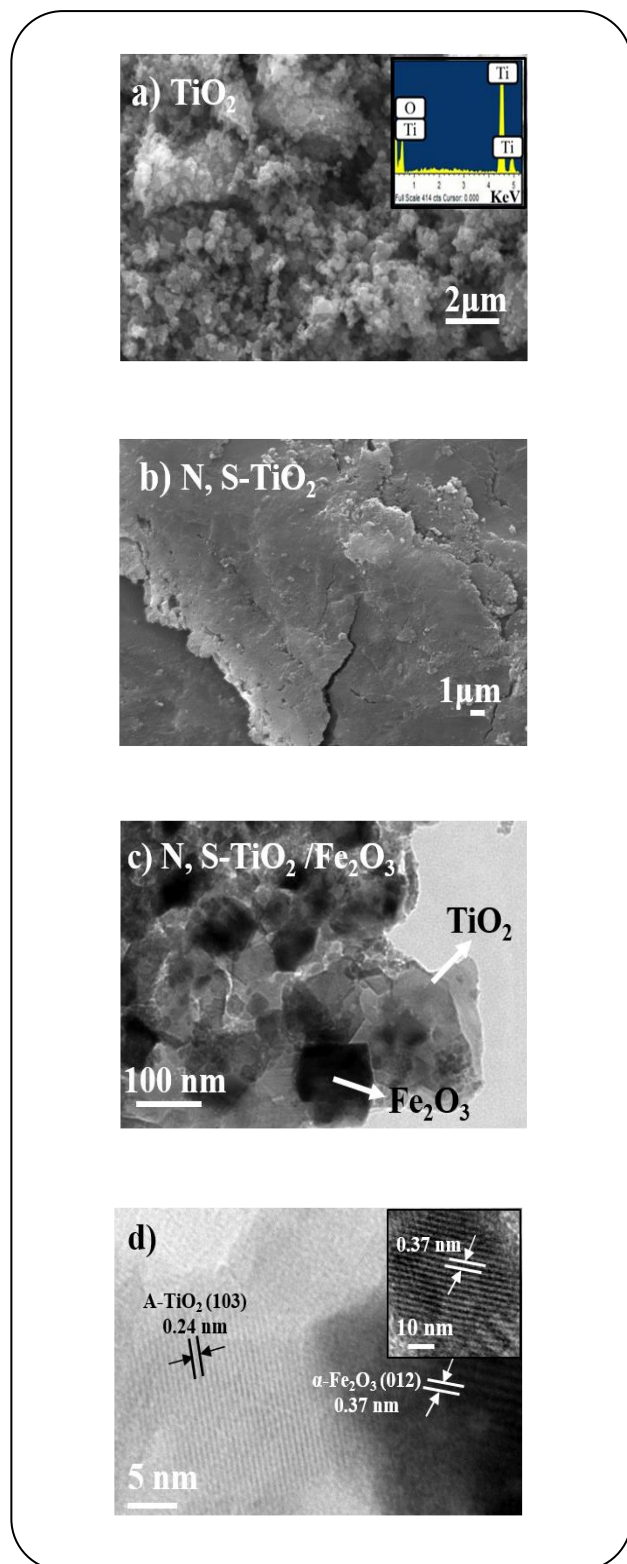


Fig. 3: FE-SEM images of (a) TiO₂ (b) N, S- TiO₂ nanoparticles (inset image shows EDS pattern); (c) TEM, and (d) HRTEM images of N, S-TiO₂/Fe₂O₃ nanoparticles (inset image shows lattice spacing).

represents to Fe-O stretching mode, the band at 3452 cm⁻¹ were assigned to O-H stretching vibrations of surface water molecules and the band at 1834 cm⁻¹ represents C-O stretching vibrations of α-Fe₂O₃ from Fig. 4(e) [35].

Cyclic voltammetry studies

Cyclic voltammetry (CV) analysis was carried out using potentiostat (K-Lyte 1.0) before and after dye degradation as shown in Fig. 5. The potentiostat is a three electrode assembly consisting of fabricated electrodes (N,S-TiO₂/Fe₂O₃ @Ti) as working electrode, Ag/AgCl as a reference electrode and Platinum mesh as counter electrode with 1M Na₂SO₄ as supporting electrolyte. All potentials were reported w.r.t Ag/AgCl reference electrode. The crystal violet dye concentration of 55mg/L was chosen for all voltammetric experiments and controlled current density of 0.1 mA/cm² was used. Cyclic voltammograms exhibit oxidation peak during forward scan and reduction peak during the reverse scan of electrolysis process. The decrease in anodic and cathodic peak currents were observed along with the time of electrolysis which indicates the oxidation of reactive azo groups [36]. The high peak current in initial CV indicates the presence of more number of reactive groups in dye molecules that leads to the higher oxidation of groups. The decrease in peak current after complete degradation indicates the reduction of reactive groups. Chrono-amperometric studies was performed at a fixed voltage (i.e. voltage where oxidation peak rises in CV) 1.2V vs Ag/AgCl for Ti/N, S-TiO₂/Fe₂O₃ at a scan rate of 0.05V/s.

UV- Visible spectra analysis

UV-Vis analysis was used to study the electrochemical degradation of crystal violet dye using UV-Vis spectrophotometer. The intense peak of crystal violet dye absorption band was observed at λ_{max} = 576 nm and decreases with the enhancement of electrochemical treatment time. The absorption becomes zero at λ_{max} = 576nm indicates the reduction of auxochrome groups-N (CH₃) responsible for the color of crystal violet dye.

The degradation rate varies with varying of electrode assemblies. It is due to the change in an electrochemical active surface area which is responsible for adsorption of more dye molecules that interact with oxidizing agents easily. Complete decolorization can be achieved by performing electrolysis using Ti/TiO₂ and Ti/N, S-TiO₂/Fe₂O₃ electrodes in 12 h and 9 h obtained from

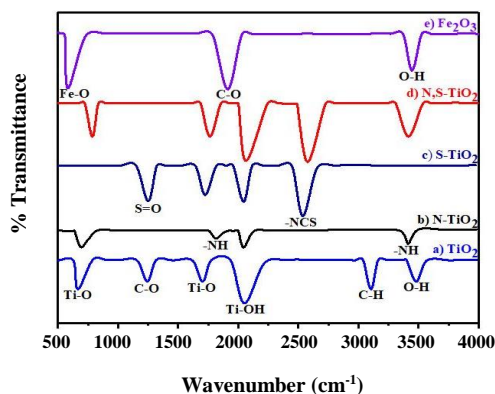


Fig. 4: FT-IR images of (a) TiO_2 (b) N-TiO_2 (c) S-TiO_2 (d) N,S-TiO_2 (e) Fe_2O_3 nanoparticles.

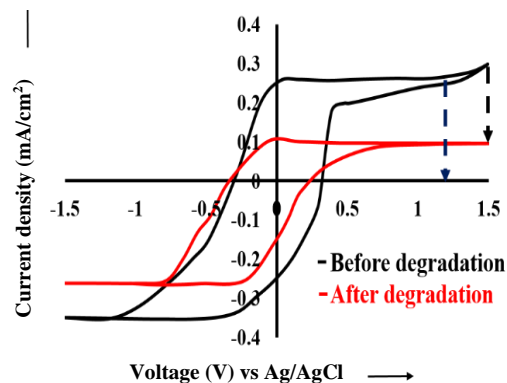


Fig. 5: Cyclic voltammetry studies of dye degradation at $\text{Ti/N,S-TiO}_2/\text{Fe}_2\text{O}_3$ (solid line - voltage where peak rises, dotted line - decrease in peak current).

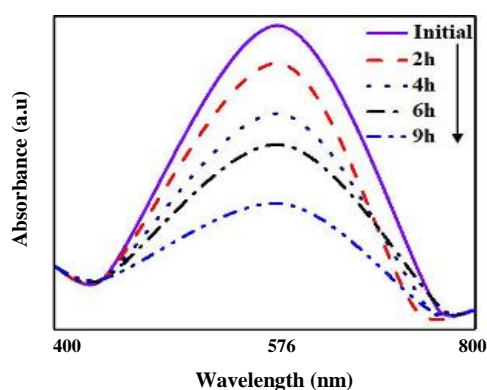
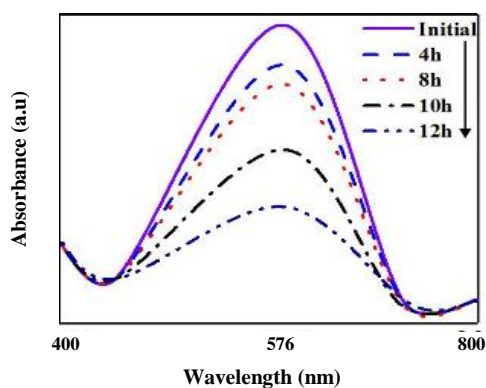


Fig. 6: Absorbance vs wavelength for a) TiO_2 b) $\text{N,S-TiO}_2/\text{Fe}_2\text{O}_3$ @ Ti substrate.

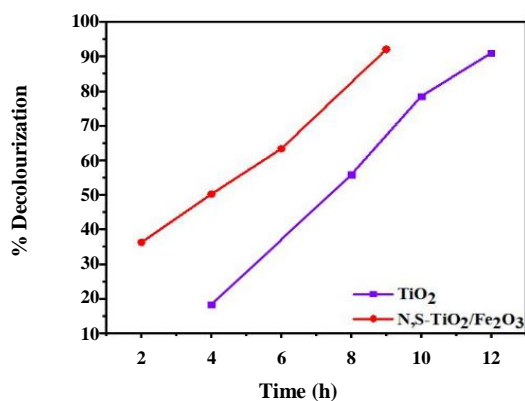


Fig. 7: Decolourization (%) vs Time (h) for TiO_2 , and $\text{N,S-TiO}_2/\text{Fe}_2\text{O}_3$ @ Ti substrate.

UV-Vis spectra shown in Fig. 6. Hence it is concluded that $\text{N,S-TiO}_2/\text{Fe}_2\text{O}_3$ achieved faster degradation compared

to the individual electrodes due to the interface of semiconductor material having suitable band edge positions and form Type-I heterojunction that favors charge carrier flow path. Therefore the doped metal oxide Heterostructure assemblies confirm the importance of charge carriers to achieve faster degradation and thereby enhancing degradation efficiency. Further, the decolourization efficiency at different time intervals can be calculated using formula

$$\% \text{ decolourization} = (A_0 - A) / A_0 \times 100 \quad (2)$$

Where A_0 and A are the absorbance (at the maximum wavelength, λ_{max}) of dye solution before/initial (at $t = 0$) and after electrochemical treatment (at $t > 0$). The decolourization percentage (%) of crystal violet dye using Ti/TiO_2 and $\text{N,S-TiO}_2/\text{Fe}_2\text{O}_3$ was achieved about 91% and 92% in 12 h and 9h as shown in Fig. 7.

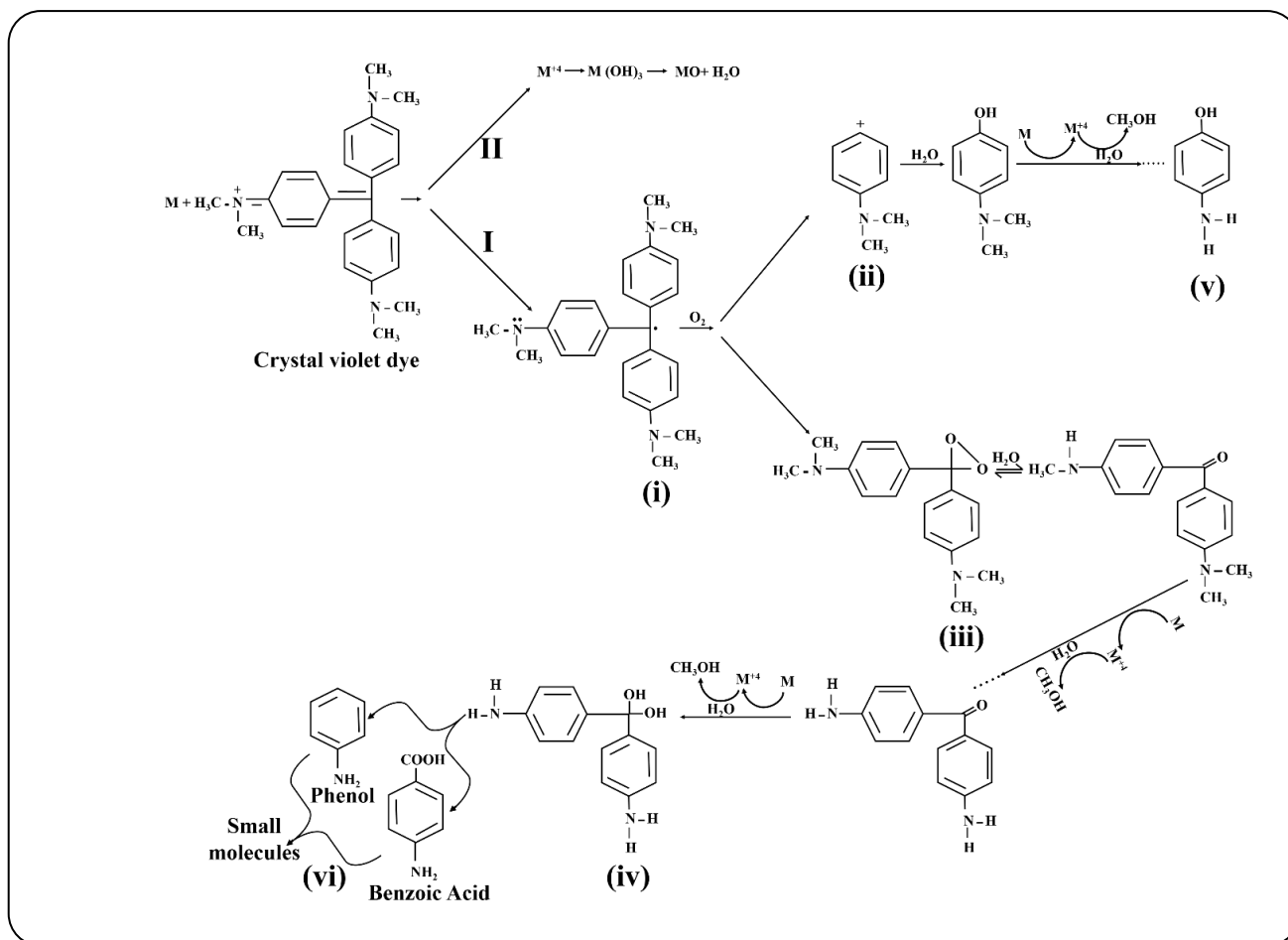
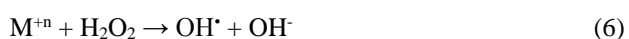


Fig. 8: Mechanism of Dye Degradation.

Mechanism of crystal violet dye degradation

Based on the literature studies an initial degradation pathway of crystal violet dye was proposed. The degradation occurs through N- dealkylation process forming a radical on central nitrogen and the destruction of chromophore structure by forming a radical on central carbon [37]. During the electrochemical treatment, the following reactions take place.



The electrons generated at the anode surface through oxidation reaction (Eq. (3)) attacks the dye molecule to destruct the conjugated structure by forming the radical on centered carbon (Fig. 8(i)) as

shown in the pathway (I). The oxygen which was generated at the anode (Eq. (4)) reacts with the intermediate formed in pathway I and leads to the formation of two intermediates as shown in Fig. 8(ii) and 8(iii) and pathway II leads to formation of water molecule. Eventually, few oxygen molecules also react with electrons at the cathode surface to form hydrogen peroxide (H₂O₂) as shown in Eq. 5 which in turn used to generate OH[•] radicals at the anode surface (Eq. (6)). Further N-demethylation occurs by interacting the methyl group of dye molecule with water molecule to form N-demethylated intermediate and methanol which continues till the formation of intermediates as shown in path way of Fig. 8(iv) and 8(v). Finally, phenol and benzoic acids were formed which in turn reduced to small molecules as shown in Fig. 8(vi). The OH[•] radicals generated through indirect reaction (Eq. 6) acts as a strong oxidizing species for degradation of the dye molecules.

CONCLUSIONS

The present study describes the importance of doped metal oxide Heterostructure assemblies (N-S TiO₂/Fe₂O₃) in electrochemical degradation of crystal violet dye. The synthesized/fabricated electrode assemblies/nanoparticles were confirmed using different characterization techniques such as XRD, SEM, EDS, and FT-IR. Electrochemical measurements show that Ti/N, S-TiO₂/Fe₂O₃ heterostructured electrode shows better performance in comparison to the individual electrodes. This can be ascribed may be due to 1) increase in electrochemical active surface area (which accommodate more oxidants to interact with dye molecules), 2) increase in electronic conductivity due to proper band edge alignments or 3) both. The results of the current study will help to design low cost catalytic assemblies in degrading complex dyes of industrial importance.

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