

Preparation and Characterization of the Inter-Polymer Complexes
between Poly (allylamine hydrochloride) and Poly(vinyl sulfonic acid)
~~Electron Donor and Electron Acceptor polymers~~

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Abstract: Poly (allylamine hydrochloride) P(AlAm.HCl) is a cationic polymer with many applications because of the existing amines group. Its importance becomes evident as poly (vinyl amine) polymer cannot be ~~produced~~ prepared from its monomer via free radical polymerization. Poly(vinyl sulfonic acid) P(VSA) is an anionic polymer with many applications, such as drug delivery and DNA modification. The synthesis of a new amphiphilic interpolymer complex (IPC) ~~production~~ was the aim of the study. Aqueous solutions of P(AlAm.HCl) and P(VSA.Na) were physically mixed in variant ~~monomer~~ repeating unit concentrations that yielded water-insoluble IPCs. FT-IR was used to identify the structures ~~behavior~~ of the IPCs, and TGA was used to determine their thermal properties. XRD patterns were performed to determine the strength of the IPCs. Strong interpolymer complexes were formed in the mole ratio (0.25:1) in feed compositions approved by XRD and TGA. Scanning electron microscopy (SEM), micrographs assessed the morphology of the IPC. ~~After removing acid from IPC of deacidified P(AlAm.HCl), a physical mixture of and P(VSA) was prepared with different mole ratios of repeating units monomers. These physical IPCs are water-soluble. FT-IR and ¹H-NMR were applied to examine the obtained formation of sulfonamide groups and the structural characterization of IPCs between P(AlAm) and P(VSA) at different molar ratios. The sulfonamide group was obtained at 170°C per hour with a (1:1) molar ratio feed composition. However, a sulfonamide group was not evident even at high temperatures due to the salt in the mixture of P(AlAm.HCl) and P(VSA.Na).~~

Keywords: Poly(vinyl sulfonic acid); Poly(allylamine hydrochloride); Interpolymer complex polymer (IPC); TGA; XRD

Introduction:

The interpolymer complex (*IPC*) is formed through the interaction of the physical structure of both electron donor and electron acceptor polymers. This physical structure can take various forms such as linear, block-graft copolymers, polymer brushes, stars, liposomes, and zwitterions. Different characteristics can be created from the baseline components by utilizing hydrogen bonds between donor-acceptor units and electrostatic interactions between polyanions and polycations functional groups. This opens up the possibility of developing various functional materials. [1-3] Recently, there has been a significant interest in studying the interactions between poly(zwitterions) and a range of chemicals, low molecular weight salts, charged polyelectrolytes, and proteins. Numerous projects have been undertaken to explore the characteristics of *IPCs*. [4-10] Particularly fascinating are the zwitterionic structures found in biopolymers. Synthetic poly(zwitterions) have the potential to produce biocompatible and non-immunogenic substances that can prevent protein destruction. [3,5] These properties make them valuable in modifying medical products and advancements in membrane structure technologies.

The studies by Kudaibergenov and Nuraje [11]; Kabanov and Zezin [12]; Wen *et al.* [13]; Ma *et al.* [14]; and Zucolotto *et al.* [15] contribute to the progressive development in interpolyelectrolyte (or interpolymer) complexes (*IPC*). The polyelectrolyte– polyelectrolyte complex's behavior has been recently reviewed. [16-20] Hydrogen-bonded *IPCs* are more considered in this point of view. [21, 22] Kabanov *et al.* conducted the initial investigation on the complexation of 2-methyl-5-vinylpyridine-acrylic acid (*2M5VPy-co-AA*) with poly(acrylic acid) (*PAA*). [12] The ionic and hydrogen bonds between the *2M5VPy-co-AA* and *PAA* in the standard cooperative system have been demonstrated to play a crucial role in the formation of the polyelectrolyte-polyampholyte complex (*PPC*). In recent decades, there has been a growing interest among researchers in solid polymer electrolytes (*SPE*). [1, 2] These materials are high-quality composites called *SPEs*. They have ionic conductivity, high stability, high specific energy, and good mechanical and thermal properties. [2-4] In addition, the applications of different chemical devices, such as rechargeable batteries and electrochemical devices, have increased the importance of these materials. [5-7] *SPEs* generally have low electrical conductivity at controlled temperatures. We can increase the *SPEs'* electrical conductivity through various methods, such as *IPC* production and ionic or mineral fillers. [6, 8, 23] *IPC*

polymers are used in drug delivery, layer-by-layer film preparation [14-16], and *DNA* and *RNA* modification. [22-23] Units fashioned from polyelectrolyte complexes are created by combining oppositely charged polymers or macromolecules. [20-22] These complexes are formed through electrostatic interactions between the positively and negatively charged components, resulting in the formation of a stable structure. [24-27] Polyelectrolytes (*PEs*) with opposite charges are brought together by electrostatic forces, resulting in the formation of polyelectrolyte complexes (*PECs*). These complexes consist of hydrogen bonding structures, which are created by the combination of polymers that have units capable of accepting and donating protons. [27-34] The formation of stereo-complexes is predominantly driven by the attractive van der Waals forces, which give rise to a complex structure with a higher level of organization. [18-21] Charge-transfer complexes (*CTCs*) are formed within polymer systems that involve both electron-accepting and electron-donating components, resulting in charge-transfer interactions. [12-14]

IPCs are products of non-covalent interactions between complementary, unlike macromolecules in solutions. These complexes include *IPCs*, *PEs*, hydrogen bonding complexes (*HBC*), and (*CTCs*). [30-34] *P(AlAm.HCl)* is a commonly encountered polycationic polymer known as polyallylamine hydrochloride. This particular polymer exhibits promising potential for various applications owing to the presence of amine groups within its molecular structure. [25]

Initially, *P(AlAm.HCl)* was synthesized through irradiation. However, a subsequent method for its synthesis involved the utilization of a radical initiator in a thermal process. The instability of poly(vinylamine) (*PVA*), an amine polymer, prevents its synthesis from the corresponding monomer. [25]

As a result, *P(AlAm)*, the closest analog to *PVA*, is synthesized from its acid salts. This is primarily due to the challenges associated with allylic monomer polymerization. *P(AlAm)* is readily available in the market, but typically in low molecular weights. [25]

P(VSA) is considered the most basic form of polymeric monoprotic acid. In terms of its potential applications, the comparison between acrylic acid *P(AA)* and polyvinyl phosphonic acid reveals the extent of their usefulness and scope. [26] Despite extensive research, there remains a lack of comprehensive understanding regarding the intricacies involved in the

synthesis of this particular subject matter. [22-25] *P(VSA)* is a polymer that plays a vital role due to its sulfonic acid composition. This polymer possesses a notable benefit in the form of a substantial concentration of sulfonate groups. [26] The practical significance lies in the anionic polyelectrolyte interactions between *P(VSA)* derivatives and biomolecules. These interactions between negatively charged *P(VSA)* derivatives and biomolecules have practical implications and are of interest in various applications. [26, 32]

The scientific and industrial communities have shown a growing fascination with polyelectrolyte complexes, as reflected in the numerous articles published on this subject. [17-22] The classification of *PECs* involves categorizing them into two distinct groups: electron donors or acceptors, and a second group known as PE-surfactants (*PE-surfs*). [17-18]

Polyelectrolyte-surfactant complexes (*PE-surfs*) encompass a combination of anionic polyelectrolytes, cationic surfactants, anionic surfactants, and ionic polyelectrolytes. [22-24, 27] An aqueous solution is created by mixing two species with opposite charges, namely polyelectrolyte-polyelectrolyte or polyelectrolyte-surfactant. Despite this, numerous methods have been devised to produce polyelectrolyte complexes (*PECs*) and polyelectrolyte surfactants (*PE surf*s). [17-22] According to the review of articles, there needs to be more detailed information about the physical formation and accurate structural identification of *P(AlAm.HCl)* and polyvinyl sulfonic acid sodium *P(VSA.Na)* salt polymers. [28, 30] In different monomer-based mole ratios, we prepared an *IPC* between *P(AlAm.HCl)* and *P(VSA.Na)*. *FT-IR* and $^1\text{H-NMR}$ characterized the structural behavior of the *IPCs*, while *TGA*, *XRD*, and *SEM* were used to investigate the thermal properties and morphological behavior. Also, we prepared an *IPC* using deacidified *P(AlAm.HCl)* with *P(VSA)* and characterized by *FT-IR* and $^1\text{H-NMR}$. The power of the acidified and deacidified complexes was compared using *XRD* and *TGA*. Furthermore, we investigated the thermal properties and morphology of the *IPC* by *TGA* and *XRD*.

Experimental:

Materials

In this study, we used Allylamine (98% pure Aldrich) as a monomer and 2,2'-azo-bis(2-methyl propanediamine) dihydrochloride (Aldrich) were used as the monomer and thermal initiator respectively. *AlAm* was complexed with 37% of 12N *HCl*. The acid was procured

from the renowned Merck Company located in Darmstadt, Germany. Additionally, precipitating substances like methanol, ethanol, and organic solvents were also obtained from the same Merck Company (Darmstadt, Germany). The monomer (*VSA.Na*) used in the experiment was acquired as a 25% aqueous solution from Aldrich. [25]

AlAm Salts Preparation

12N *HCl* solution was filled in three-necked containers, which included with an attached condenser and stirrer. The temperature of *AlAm* was lowered to a range of -10 to 0°C, and subsequently, *HCl* was added drop by drop to the *AlAm* solution while maintaining the temperature of around 5-10°C. This process resulted in the formation of a clear solution of known as *AlAm.HCl*. [25]

Polymerization of *AlAm.HCl* in Water

The 6.68 g of 70 % (*AlAm.HCl*) solution was poured into a four-necked flask container, including a nitrogen inlet, stirrer, thermometer, and reflux condenser.

The process began by adding an initiator solution, specifically 2,2'-azo-bis (2-methyl propanediamine) dihydrochloride, to 0.35 mL of water. This resulting solution was then combined with the monomer and stirred for a duration of ten hours at a temperature of 50°C. After the initial ten hours of stirring, the stirring ceased, and the polymerization process continued for an additional 40 hours. The resulting polymer was precipitated in ethanol, resulting in with the formation of a viscose and colorless compound. This precipitate was then filtered, washed, and subjected to extraction with ethanol in a soxhlet apparatus for a period of 15 hours, which equated to approximately 60 cycles. The purpose of this extraction was to remove any remaining unpolymerized monomer. Finally, the precipitate was dried in a vacuum oven at 50°C, resulting in the formation of a white powder known as poly(allyl ammonium) chloride. [25]

Deacidification of *P(AlAm.HCl)*

In the methanol suspension solution, we introduced 2.9 N sodium hydroxide (*NaOH*) to the *P(AlAm.HCl)*. The suspension was allowed to stand for a duration of one hour, after which it was refrigerated overnight. The resulting precipitate of *NaCl* was separated by decanting and solution was subsequently evaporated until completely dry. [25, 27]

Polymerization of VSA in Water

A four-necked ~~container~~ flask equipped with a nitrogen inlet, stirrer, and reflux condenser was utilized to hold 6.68 g of a 25% VSA solution. The solution was subjected to deoxygenation by nitrogen purge and heated to a temperature of 50°C. Subsequently, a solution containing the ~~an~~ initiator, specifically 2,2'-azo-bis (2-methyl propanediamine) dihydrochloride, was prepared by dissolving it in 0.35 mL of water. This initiator solution was then added to the monomer. The resulting mixture was stirred for a duration of ten hours while maintaining the temperature at ~~of~~ 50°C. Following this, the stirring was halted, and the polymerization process was allowed to continue for an additional 40 hours. ~~As a result,~~ The polymer formed was precipitated in ethanol, yielding a colorless and viscous liquid product. ~~polymer.~~ The precipitate was subsequently filtered, washed, and subjected to extraction with ethanol in a soxhlet apparatus for a period of 15 hours, which involved approximately 60 cycles. This extraction process aimed to eliminate any remaining unpolymers. Finally, the precipitate was dried under vacuum conditions at ~~a temperature of~~ 50°C, resulting in the formation of a white powder of *P(VSA)*. [26]

Preparation of Interpolymer Complexes

To create *IPCs* with varying repeating unit-based mole ratios using different concentrations of polymers. ~~monomers,~~ Predetermined amounts of the salts of *P(AlAm.HCl)* and *P(VSA.Na)* were dissolved in water. The resulting solution containing both polymers was carefully mixed and then separated through precipitation using a filter. Subsequently, the ~~mixture~~ filtrate was subjected to vacuum at a temperature of 50°C for a duration of 24 hours.

Characterization of *IPCs*:

FT-IR Analyses

FT-IR spectrophotometer (Nicolet 520 model U.S.A) was ~~applied~~ used for *FT-IR* analysis. The KBr discs were made for the samples. Spectra conditions were 64 scans ~~and~~ at 4 cm⁻¹ resolution.

The *FT-IR* ~~spectroscopy~~ bands for *P(AlAm.HCl)*;

3300-3500 (γNH_3^+), 2800-3000 (γCH_2), 1500-1700(NH_3^+ bending) 1000-1200 (NH_3^+ bending), 700-900(NH_3^+ bending). [25]

For *P(AlAm)*;

3300 (γNH_2), 2800 (γCH_2), 1500-1600(NH_2 bending), 700-900(NH_2 bending), 1350-1450(CH bending). [25]

For *P(VSA.Na)*, and *P(VSA)*;

The spectrum displays several characteristic peaks at different wavenumbers. These include peaks at 1037 cm^{-1} and 1120 cm^{-1} , which correspond to the stretching vibrations of the sulfur-oxygen bond ($\gamma\text{S}=\text{O}$). Another peak is observed at 3500 cm^{-1} , indicating the presence of hydroxyl groups ($\gamma\text{O}-\text{H}$). The range between $3000\text{-}2900\text{ cm}^{-1}$ shows peaks related to the stretching vibrations of carbon-hydrogen bonds ($\gamma\text{C}-\text{H}_2$). Additionally, there are peaks at 1700 cm^{-1} (CH_2 bending), 1650 cm^{-1} (SO_2 bending), 1400 cm^{-1} (CH bending), 1200 cm^{-1} (OH bending), 1050 cm^{-1} (SO_2 bending), and $500\text{-}800\text{ cm}^{-1}$ (SO_2 bending). These peaks provide valuable information about the molecular structure and functional groups present in the compound. [25]

~~The IPC was prepared by dissolving and mixing *P(AlAm.HCl)* and *P(VSA.Na)* in water while continuously stirring. The resulting precipitant was then filtered, dried, and subsequently placed in a vacuum at a temperature of $50\text{ }^\circ\text{C}$ for a duration of 24 hours. FT-IR analysis showed the peak intensities at $1650\text{-}2000\text{ cm}^{-1}$ and 3000 cm^{-1} . It is thought that peak intensities were increased with increasing the amount of *P(AlAm.HCl)*.~~

NMR analyses

A 0.01 g portion of the polymer sample was dissolved in 0.7 mL of deuterium oxide (D_2O), and subsequently, a study using a $^1\text{H-NMR}$ technique was conducted. The $^1\text{H-NMR}$ instrument employed for this study was the Bruker U.S.A Ultra Shield 400 MHz NMR spectrometer.

$^1\text{H-NMR}$ peaks for *P(AlAm.HCl)*;

0.9-1.2 ppm CH_2 polymerization grows allyl degradation, 1.2-1.6 ppm CH_2 main chain normal polymerization, 1.8-2.2 ppm CH main chain normal polymerization, 2.8-3.3 ppm CH_2 side chain normal polymerization, 3.5-3.7 CH_2 side chain polymerization grow allyl degradation. [25]

¹H-NMR for P(AlAm);

0.9-1.4 ppm CH₂ main chain polymerization, 1.4-1.7 ppm CH main chain polymerization, 2.4-3.1 ppm CH₂ side chain, 3.5-3.8 ppm NH₂. [25]

¹H-NMR for P(VSA);

1.4-3.0 ppm CH₂ main chain, 3.0-4.0 ppm CH main chain, 5.4-6.2 ppm OH. [26]

¹H-NMR for P(VSA.Na);

1.4-2.8 ppm CH₂ main chain, 3.0-4.0 ppm CH main chain, 5.4-6.2 ppm OH. [26]

¹H-NMR for P(AlAm)-P(VSA) (1:1) monomer repeating unit;

1.2-1.8 ppm CH₂ AlAm in the main chain, 1.8-3.0 ppm CH₂ main chain in VSA, 3.0-4.4 ppm CH main chain VSA, 2.4-3.0 ppm CH₂ side chain in AlAm, 1.4-1.7 ppm CH main chain in AlAm, 5.7-6.4 ppm OH in VSA

¹H-NMR for (1:0.25) monomer repeating unit;

1.2-1.8 ppm CH₂ AlAm main chain, 1.4-1.7 ppm CH AlAm main chain, 1.9-3.1 ppm CH₂ main chain VSA, 3.2-4.4 ppm CH main chain VSA, 2.5-3.0 ppm CH₂ side chain AlAm, 5.7-6.5 ppm OH VSA

¹H-NMR for (0.25:1) monomer repeating unit;

1.9-3.0 ppm CH₂ main chain VSA, 1.8-2.0 ppm CH₂ main chain AlAm, 1.4-1.7 ppm AlAm, 2.5-3.0 ppm CH₂ side chain AlAm, 3.0-4.0 ppm CH main chain VSA, 5.7-6.5 ppm OH VSA

Investigation of the Thermal Properties

IPC samples were characterized by conducting thermal gravimetric analysis (TGA) using a Perkin Elmer Pyris model thermogravimetric analyzer from the United States. The samples under investigation consisted of 5-10 mg of powders, which were subjected to TGA analysis. The nitrogen flow rate was maintained at 20 mL/min during the experiments. The weight loss as a function of temperature ratio was measured as the temperature varied from room temperature to 900 °C at a heating rate of 20 °C/min. Subsequently, the collected data were subjected to analysis.

XRD Analysis

To assess the crystallinity of *IPC*, a Rigaku *D/Max 2200* diffractometer from Japan was employed to analyze 10 mg of powder. The analysis encompassed the utilization of $CuK\alpha$ ($\lambda = 1.54059 \text{ \AA}$) radiation and measured the angle 2θ while maintaining a temperature range of 0°C to 60°C .

SEM Analyses

The *XRD* analysis samples were subjected to a treatment involving the use of two milligrams of *IPC* powder, which was subsequently coated with gold using a vacuum evaporator. The morphologies of the *IPC* were then examined using a Jeol *JSM-6060LV* model Scanning Electron Microscope (*SEM*) from the U.S.A.

Result and Discussion:

P(AlAm.HCl) is cationic, and *P(VSA)* is an ionic polymer. These polymers, besides as well as their copolymers, were synthesized by Sepehrianazar *et al.* [25,27]. In this study, we investigate *IPC* formation between these two polymers.

Investigation of IPCs Structural behavior by FT-IR

To compare *AlAm.HCl/VSA* co-polymers and their *IPCs*, a mixture of different mole ratios of *P(AlAm.HCl)* and *P(VSA.Na)* were prepared in an aqueous solution to form an *IPC*. The resulting *IPC* was obtained by mixing *P(AlAm.HCl)* with *P(VSA.Na)*. The *FT-IR* spectra analysis revealed that the NH_2 stretching intensity and the amount of *AlAm* increased at 3100 cm^{-1} . Additionally, the bending intensity of $C-H$ and $N-H_2$ also increased at 1500 cm^{-1} . The power of $S-O_2$ bending, similar to the amount of *P(VSA.Na)*, increased at 1700 cm^{-1} . Furthermore, the $O-H$ bending intensity showed an increase at 1200 cm^{-1} , as depicted in Fig1.

In *P(VSA.Na)*, SO_2 bending created the peak intensities at 1050 cm^{-1} . Developed bending at 1300 cm^{-1} was because of OH bending, and OH stretching peaked at 3500 cm^{-1} . An increase in the *P(VSA.Na)* amount can make prominent peaks. *FT-IR* spectra were obtained after heating the polymer at 170°C for 30, 60, and 180 minutes. All obtained spectra were the same with no difference (Fig.1-2).

The bending vibrations of OH , SO_2 , and NH_2 were observed at 1300 cm^{-1} , 1150 cm^{-1} , and 1450 cm^{-1} , respectively. The interaction between $P(AlAm.HCl)$ and $P(VSA.Na)$ IPC (Fig. 1) resulted in a displacement towards longer wavelengths. Although the $FT-IR$ spectrum of $P(AlAm.HCl)$ and $P(VSA)$ has been examined and documented, it is not included in this particular listing.

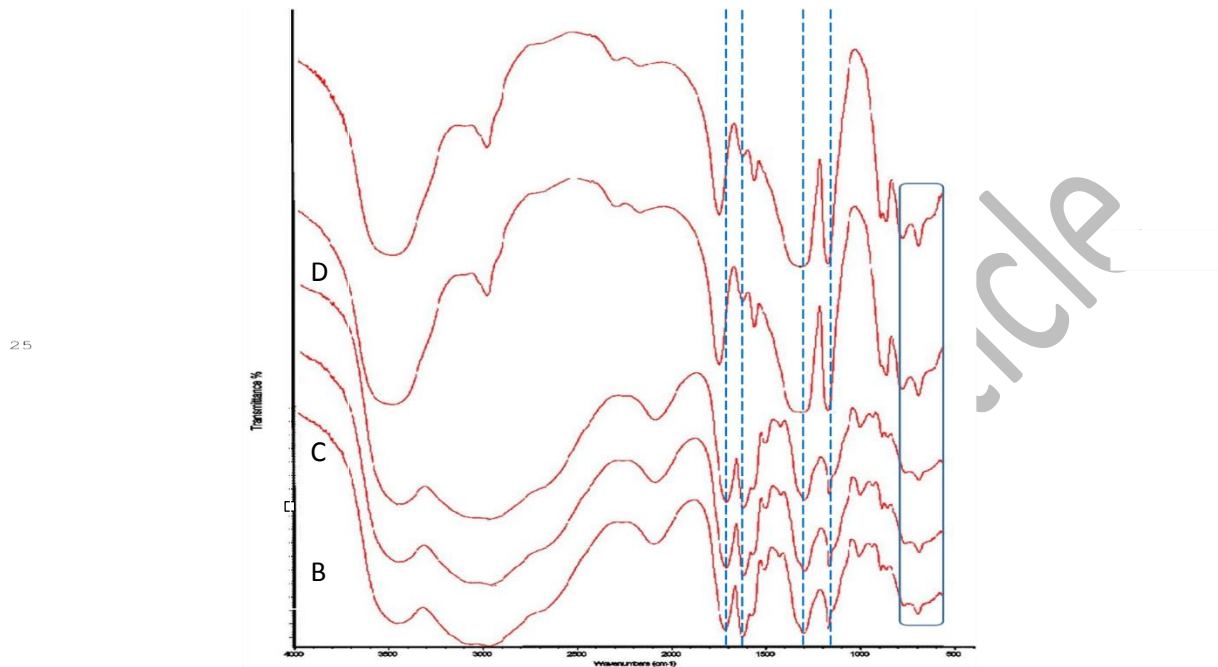


Figure 1. $FT-IR$ spectra of $P(AlAm.HCl)$ and $P(VSA.Na)$ IPCs, prepared in different mole ratios of feed compositions **a**) (1:1), **b**) (1:0.5), **c**) (1:0.25), **d**) (0.5:1), **e**) (0.25:1)

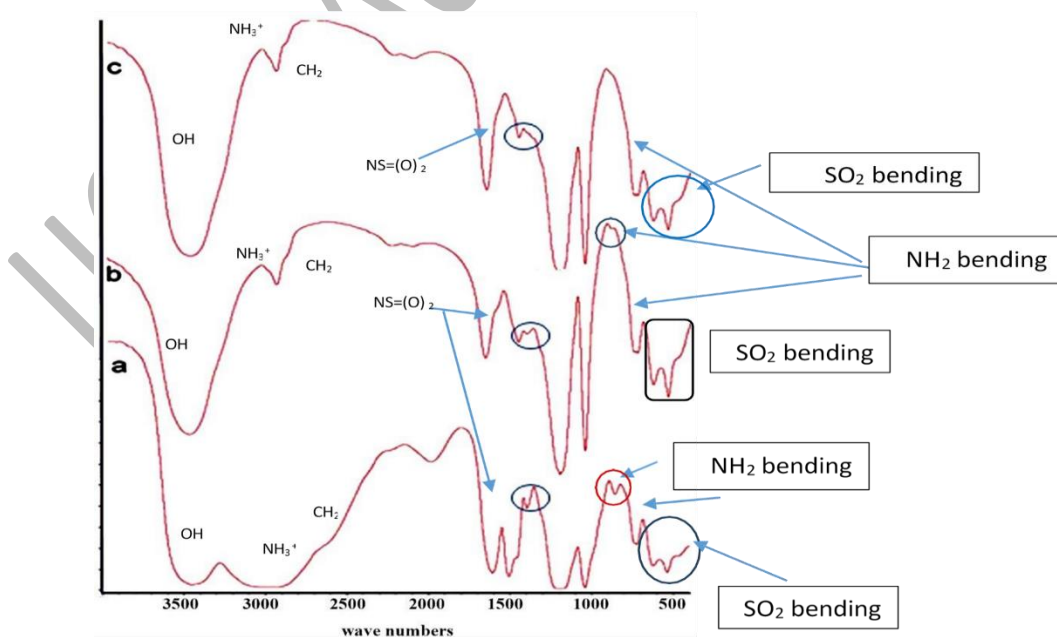


Figure 2. $FT-IR$ spectra of $P(AlAm)$ and $P(VSA)$ in different mole ratio of feed composition **a**) (1:1), **b**) (1:0.25), **c**) (0.25:1)

The mixture of $P(AlAm.HCl)+P(VSA.Na)$ was subjected to a process to prepare IPC with $P(AlAm)$ and $P(VSA)$. This involved the removal of HCl from the mixture, resulting in the conversion of $P(AlAm.HCl)$ to $P(VSA)$ and the dissolution of polymers in water. Subsequently, $P(VSA)$ was added to $P(AlAm)$, leading to the occurrence of a white precipitate. The precipitate was then dried in a vacuum at $50^{\circ}C$ for 24 hours. To analyze the composition of the resulting IPC , $FT-IR$ spectra were obtained and are presented in Figure 3. Notably, the stretching of NH_2 was observed at 3100 cm^{-1} , indicating a (1:1) mole ratio of feed composition. Sulfoamidization occurred at 1700 cm^{-1} in (0.25: 1) and (1:0.25) mole ratios of feed composition. Weak NH_2 bending was shown at 900 and 1400 cm^{-1} . $IPCs$ were prepared at different mole ratios of feed composition and were heated at $170^{\circ}C$ for 30, 60, and 180 minutes. $FT-IR$ spectra of $IPCs$ are shown in Figure 3. The sulfonamide group is observed at $1690-1750\text{ cm}^{-1}$, secondary to the temperature effect in $FT-IR$ spectra. Increased peak intensities observed at $1690-1750\text{ cm}^{-1}$ support the sulfonamide bond.

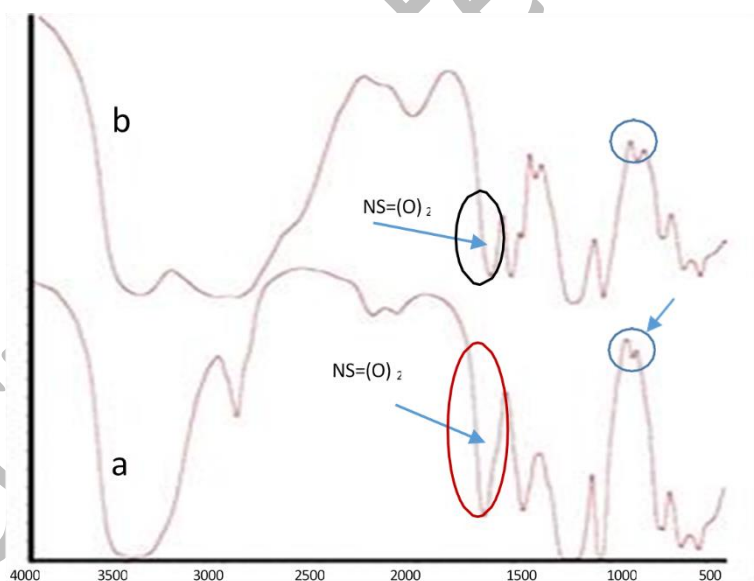


Figure 3. $FT-IR$ spectra of $P(AlAm)+P(VSA)$ IPC at **a)** $170^{\circ}C$, **b)** room temperature

Table1: *FT-IR* spectra of *P(AlAm) + P(VSA)* at room temperature and at 170 °C

Bands (cm^{-1})	Assignments
3500	O-H stretch
3000	N-H ₂ stretch
2950	C-H ₂ stretch
1700	NH-SO ₂ in AlAm and VSA
1500	N-H bending in AlAm
1400	N-H ₂ bending
1200	O-H bending
1000	C-H bending
500-700	N-H and S-O ₂ bending

Figure 3 shows the *FT-IR* spectra analysis at 170°C and 25°C. The shoulder bending of *NH*₂ disappeared at 1400 cm^{-1} in 25°C but increased at 170 °C. The increasing peak intensity indicates sulfamidization at 1690-1750 cm^{-1} .

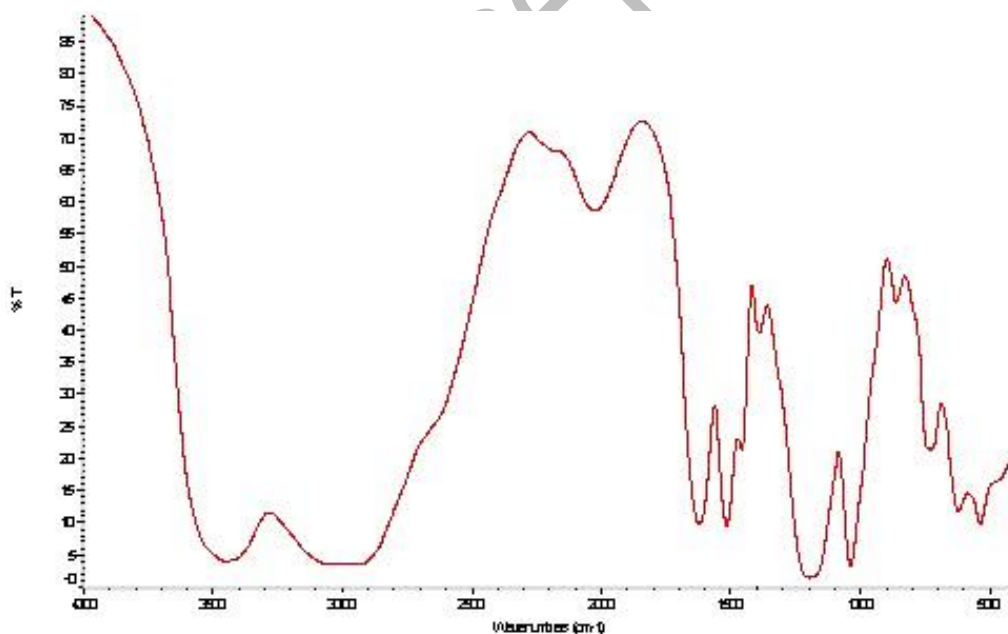


Figure 4. *FT-IR* spectrum of *IPC* of (*AlAm.HCl*) + *P(VSA.Na)* in (1:1) mole ratio of feed composition at room temperature

By taking the (1:1) mole ratio of feed composition of the $P(VSA.Na) + P(AlAm.HCl)$ IPC, no variation was shown at room temperature and 170°C. This is the hindrance of the sulfonamide reaction in the presence of $NaCl$ (Fig. 4)

Investigation of IPC Structure by NMR

Different mole ratios of $P(AlAm)$ and $P(VSA.Na)$ were used to prepare IPCs in the feed composition. The NMR spectra of the IPCs confirmed the formation of weak structure complexes. $P(AlAm)$ and $P(VSA.Na)$ were easily dissolved in deionized water (D_2O). The structural behavior of the IPCs was investigated through 1H -NMR measurements of the homopolymer, as shown in Figure 5-6. In the 1H -NMR spectrum of $P(AlAm.HCl)$, the presence of the CH_2 peak at one ppm indicated degradative chain transfer on the main chain. The CH_2 peak also exhibited a rise due to main chain polymerization at 1.2-1.7 ppm. The CH peak, observed at 1.8-2.2 ppm, was attributed to polymerization. The height of the CH_2 proton at 2.8-3.3 ppm indicated side chain transfer, while the CH_2 peak at 3.4-3.7 ppm was due to degradative chain transfer on the side chain. Figure 6b depicted the 1H -NMR spectrum of $P(AlAm)$, which showed the presence of the CH_2 peak (0.99-1.4 ppm), CH peak (1.4-1.7 ppm), CH_2 peak on the side chain (2.4-2.8 ppm), NH_2 peak (3.4-3.8 ppm), and CH_2 peak due to degradative chain transfer on the side chain (2.1-2.2 ppm). Upon acid removal, the CH_2 group of the primary chain shifted to 0.98 ppm, CH shifted to 1.4 ppm, CH_2 on the side chain shifted to 2.4 ppm, and the free amine group shifted to 3.4 ppm. These shift values indicated the removal of acid from the $P(AlAm.HCl)$ complex. [25]

δ 1H -NMR 400MHZ (D_2O): 4.2-2.9 (CH_2CHSO_3Na), polymer backbone, 2.9-1.4 (CH_2CHSO_3Na), polymer backbone, 1.45 (2H, OCH_2CH_3), 1.45-1.2 (3H, $CHCH_3$). [26-27]

δ ^{13}C -NMR 400 MHZ (D_2O): 38-28 ppm (CH_2CHSO_3Na), polymer backbone, 60-50 ppm (CH_2CHSO_3Na), polymer backbone. [26-27]

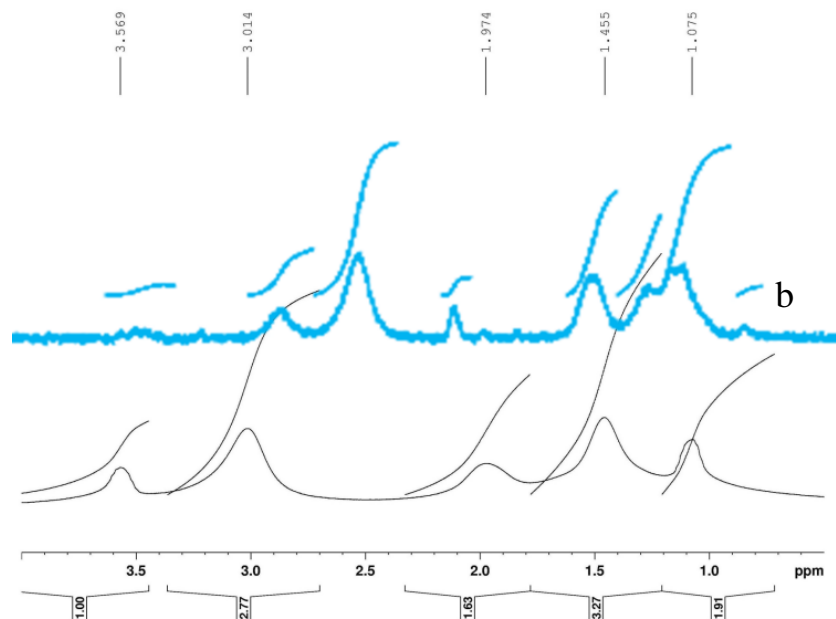


Figure 5. The $^1\text{H-NMR}$ spectrum of **a)** $P(\text{AlAm.HCl})$, **b)** $P(\text{AlAm})$

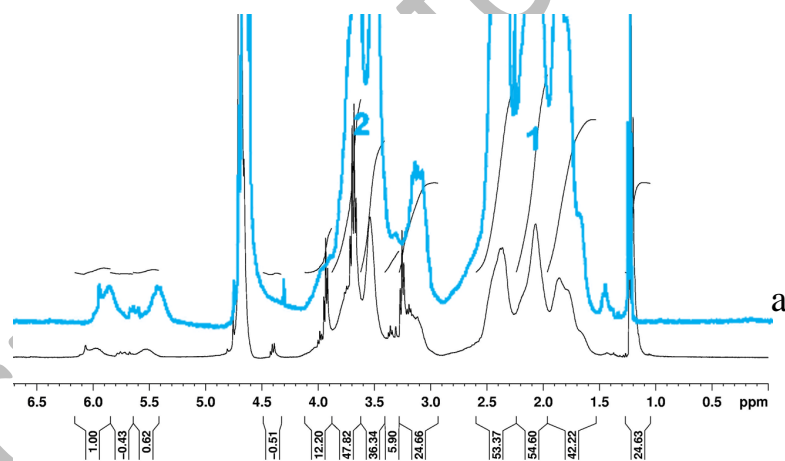


Figure 6. The $^1\text{H-NMR}$ spectrum of **a)** $P(\text{VSA})$, **b)** $P(\text{VSA.Na})$

The complexation between $P(\text{AlAm})+P(\text{VSA})$ was evaluated by the $^1\text{H-NMR}$ spectroscopy. The IPCs of $P(\text{AlAm})+P(\text{VSA})$ with (1:1), (1:0.25), and (0.25:1) mole ratios on feed composition are shown in Fig 7. The chemical shifts are compatible with the IPC , $[P(\text{AlAm})+P(\text{VSA})]$ shift to a high field compared to the pure homopolymers of $P(\text{AlAm})$ and $P(\text{VSA})$ secondary to the IPCs complexation. The $^1\text{H-NMR}$ scope of the (AlAm.HCl) , $P(\text{AlAm})$, $P(\text{VSA})$, and $P(\text{VSA.Na})$ homopolymers are shown in Fig. 5-6.

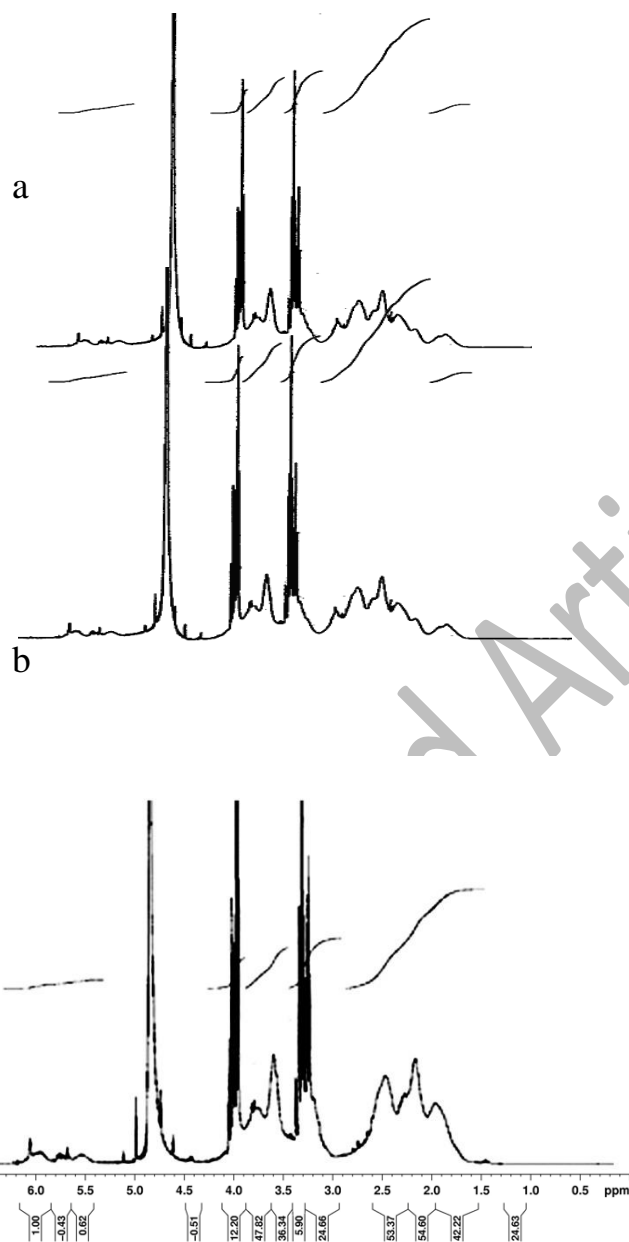


Figure 7. **a)** $^1\text{H-NMR}$ prepared (1:1) mole ratio of $P(\text{AlAm})+P(\text{VSA})$ on a repeating unit basis, **b)** (1:0.25) mole ratio of $P(\text{AlAm})+P(\text{VSA})$ on a repeating unit basis, **c)** (0.25:1) mole ratios of $P(\text{AlAm})+P(\text{VSA})$ on a repeating unit basis.

Investigation of IPC by XRD and SEM

XRD and SEM investigation of $P(\text{AlAm.HCl})+P(\text{VSA.Na})$

The X-ray diffraction (XRD) patterns of both homo- and copolymers were obtained using a Rigaku D/MAX-2200 diffractometer. Despite the amorphous nature of the structures, the XRD patterns of $P(\text{AlAm})$ exhibited sharp and well-defined peaks. This phenomenon can be attributed to the removal of HCl from $P(\text{AlAm})$ and the presence of NaCl in the medium, which

prompted the conversion of the structure into a more standardized form. (Fig. 8). The *XRD* patterns of *P(AlAm.HCl)* and *P(VSA.Na)* polymers exhibited well-defined peaks, which were attributed to the presence of *NaCl* in the medium and the considerable degree of complexation. These sharp peaks were observed across various mole ratios utilized in the preparation of the compositions. (Fig. 9). The *XRD* profiles of the *IPCs* consisting of acidless *P(AlAm)* and exhibiting a highly weak complexation and amorphous structure were examined. (Fig. 11)

Upon removal of the acid, the formation of *P(VSA) IPC* from *P(AlAm)* occurred, resulting in an observed balance reaction. This reaction was subsequently observed in *IPCs*. To determine the degree of crystallinity, it is necessary to separate the scattering from the crystalline and amorphous components in the diffraction pattern. The *XRD* patterns of *P(AlAm.HCl)* and *P(VSA.Na)* were recorded separately, revealing the presence of only amorphous peaks. To calculate the crystallinity of the *P(AlAm.HCl)* and *P(VSA.Na) IPCs*, the following equations were utilized. The degree of crystallinity (x_c) is determined by the ratio of the integrated crystalline scattering to the total distribution, encompassing both crystalline and amorphous components, as expressed in Eq. (1).

$$\text{Eq. (1)} \quad x_c = \int_0^{\infty} s^2 I_c(s) ds / \int_0^{\infty} s^2 I(s) ds$$

The magnitude of the reciprocal-lattice vector, denoted as s , can be calculated using the formula $s = (2\sin q)/l$, where q represents the one-half angle deviation of diffracted rays from incident X-rays, and l represents the wavelength. The intensities of coherent X-ray scattering from both the crystalline and amorphous regions are represented by $I(s)$ and $I_c(s)$, respectively. Additionally, $I_c(s)$ specifically refers to the intensity of the crystalline region in the polymer sample. The X-ray diffraction (*XRD*) measurements were used to determine the basal spacing (d_{001}) of the interpenetrating polymer composites (*IPCs*). The *XRD* patterns, which depict the crystallinity of the power complexation process, are presented in Figures 9-12. All the *XRD* parameters obtained from the analysis are summarized in Table 2. Additionally, the X-ray diffraction diagrams of the *P(AlAm.HCl)+P(AlAm)* powders can be observed in Figures 8-9.

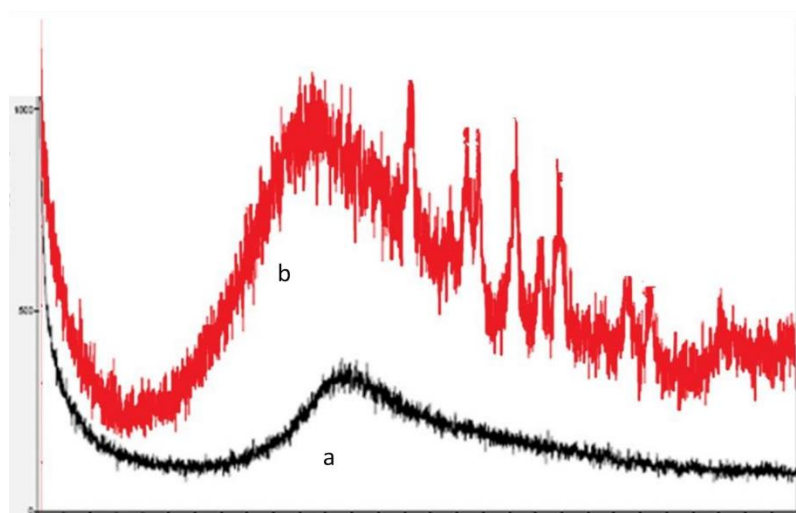


Figure 8. XRD pattern of a) *P(AlAm.HCl)* and b) *P(AlAm)*

It is observed in Table 2 that the *P(AlAm.HCl)* amount is low, while the crystallinity of the *IPC* (0.25:1) mole ratio of a repeating unit is high. It means the most potent complexation was obtained.

Table 2. XRD parameters and crystallinity of *P(AlAm.HCl)+P(VSA.Na)* IPCs

<i>P(AlAm.HCl)+P(VSA.Na)</i> Mole ratio	2θ ($^{\circ}$)	Intensity (Counts)	Peak area (<i>S</i>)		χ_c (%)
			<i>S_a</i>	<i>S_c</i>	
(1:1)	31.6	875	3935	2776	41.4
(1:0.5)	32.082	1105	3437	2676	43.8
(1:0.25)	31.841	744	3864	2663	40.8
(0.5:1)	31.841	1620	4623	3854	45.5
(0.25:1)	31.738	2696	4808	4715	49.5

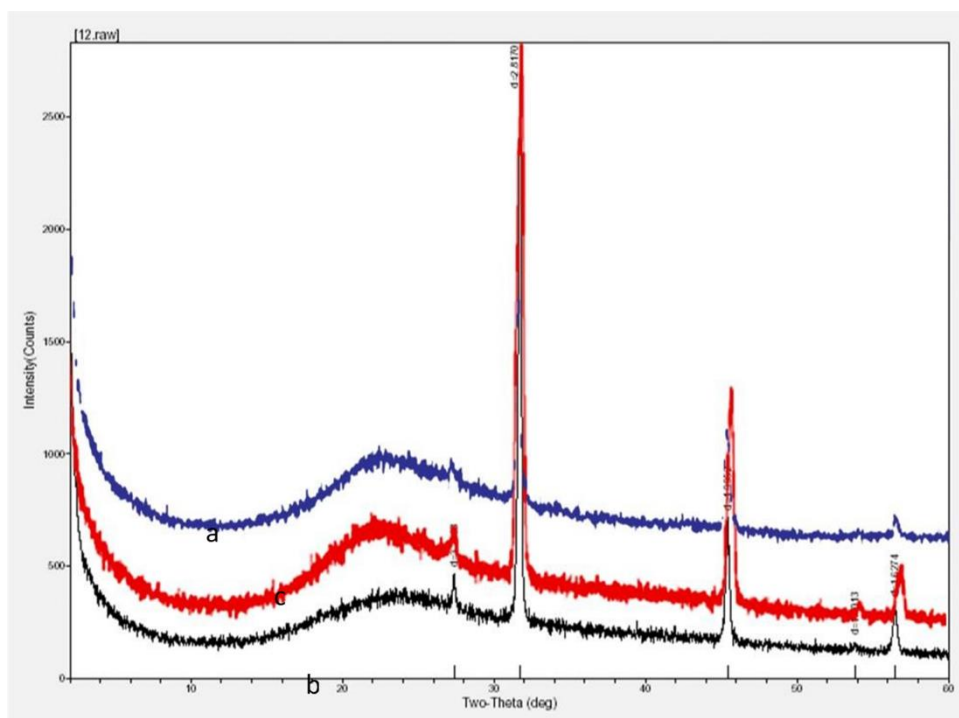


Figure 9. XRD patterns of IPCs between $P(AlAm.HCl)$ and $P(VSA.Na)$ a) (1:1), b) (0.5:1), c) (0.25:1) mole ratios on a repeating unit

For the investigation of morphology behavior, we used *SEM*. The morphology of *IPC* in (0.25:1) mole ratio of a repeating unit $P(AlAm.HCl)+P(VSA.Na)$ shows the $NaCl$ crystal on the *IPC*. (Fig10) It was the most potent *IPC* that was confirmed by *XRD* analysis.

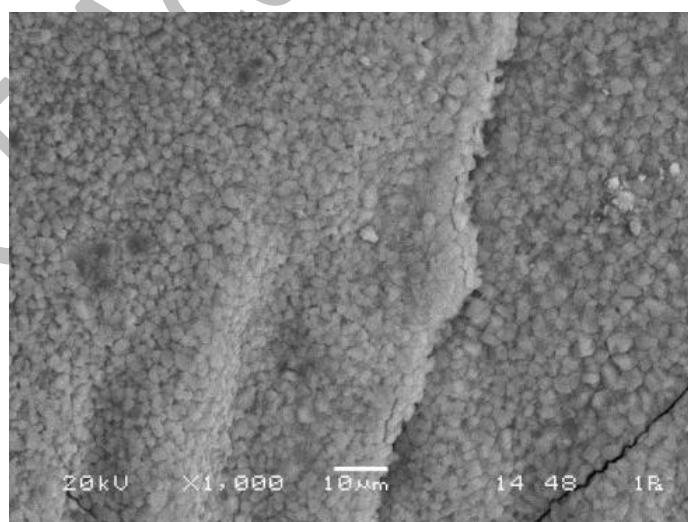


Figure 10. *SEM* micrograph of IPCs between $P(AlAm.HCl)$ and $P(VSA.Na)$ on (0.25:1) mole ratio on a repeating unit

Fig. 11 presents the preparation of $P(AlAm.HCl)+P(VSA.Na)$ *IPC*s through the physical mixture method at various mole ratios, with the monomer unit serving as the reference point.

The XRD patterns were obtained for the five different compositions of $P(AlAm.HCl)+P(VSA.Na)$ IPCs. Notably, the XRD patterns exhibited prominent peaks corresponding to $NaCl$, indicating the formation of a robust complexation resulting from the $NaCl$ formation.

Investigation of $P(AlAm)+P(VSA.Na)$ IPCs by XRD

XRD patterns of $P(AlAm)+P(VSA.Na)$ IPCs had amorphous peaks.

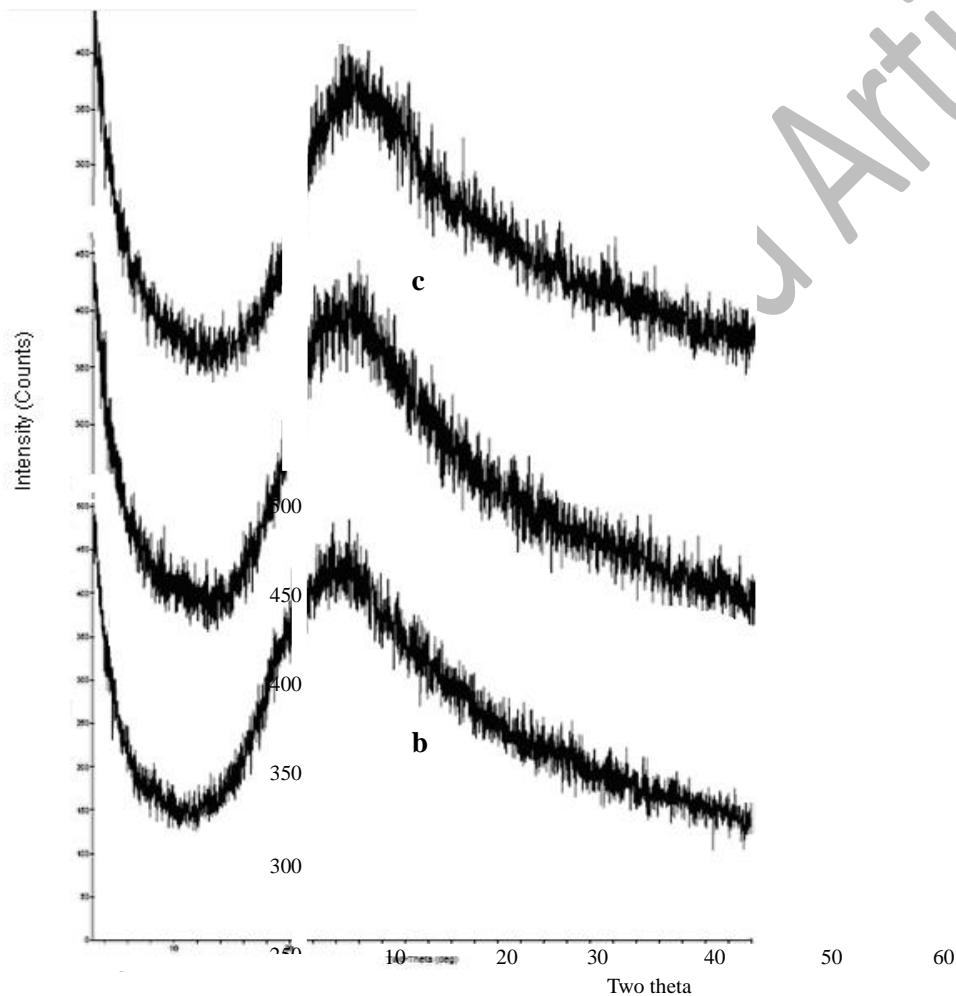


Figure 11. XRD patterns of **a)** (1:0.25), **b)** (1:0.5), **c)** (1:1) mole ratio of feed compositions $P(AlAm)+P(VSA.Na)$ IPCs

In accordance with Figure 11, the removal of HCl from the strong complexation resulted in a decrease in crystallinity, leading to the complete amorphous nature of the structures. [25-27]

Investigation of Thermal Behavior of $P(AlAm.HCl)+P(VSA.Na)$ IPCs by TGA

The TGA technique was employed to examine the thermal stability of $P(AlAm)$ and $P(AlAm.HCl)$ to investigate their behavior under varying temperatures. The thermal properties of $P(AlAm.HCl)$ and $P(AlAm)$ are illustrated in Fig 12a and Fig 12b, respectively. These figures provide valuable insights into the thermal characteristics of both compounds, shedding light on their stability and potential applications.

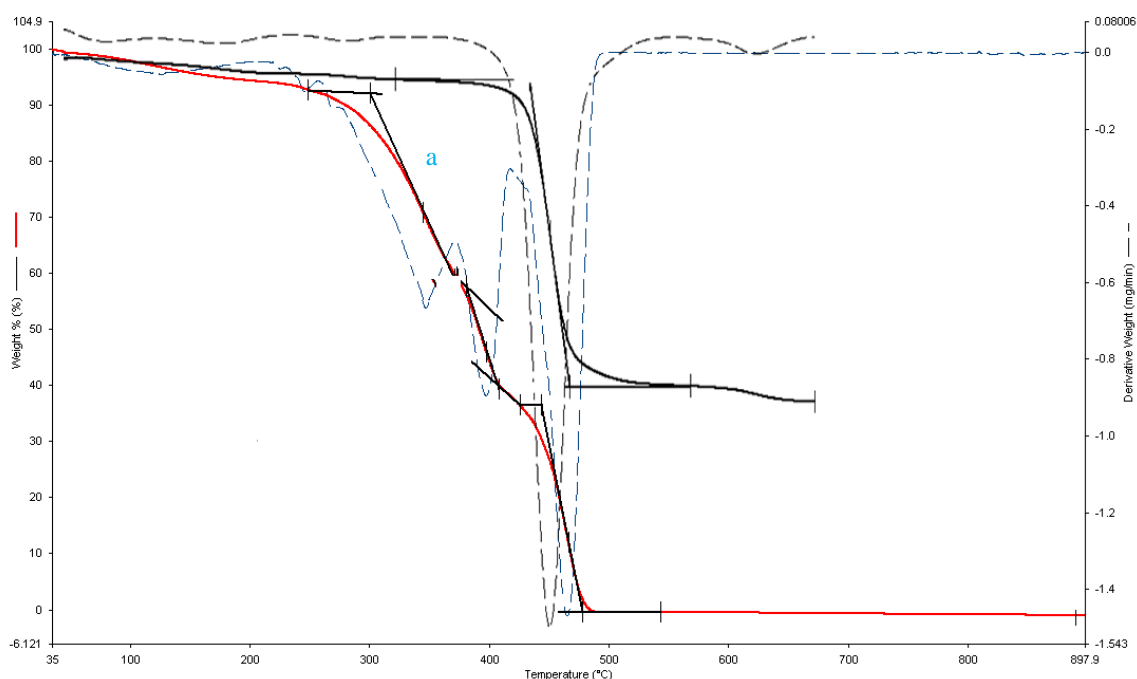


Figure 12. Thermal degradation of **a)** $P(AlAm.HCl)$, **b)** $P(AlAm)$

In Figure 12a, it can be observed that the thermal degradation of $P(AlAm.HCl)$ takes place in three distinct steps. The initial step occurs between 300°C and 370°C, during which HCl is removed from the base polymer. The second step, which takes place from 380°C to 408°C, corresponds to the removal of the amine groups. The weight loss percentage observed during these two steps corresponds to the weight of HCl and the amine groups. The third step begins at 444°C and concludes at 478°C, during which all of the polymers decompose and transform into carbon black. [25]

On the other hand, the thermal behavior of *P(AlAm)* demonstrates a strong thermal property, as depicted in Figure 12b. This property becomes evident at approximately 440°C. The weight loss of the polymer occurs in a single step, starting at 444°C and ending at 479°C. Interestingly, a significant portion of the polymer, approximately 45%, remains thermally stable even at temperatures as high as 700°C. This enhanced stability at elevated temperatures can be attributed to the formation of ring structures along the polymer chain. These cyclic structures exhibit stability during the thermal degradation process of *P(AlAm)* under a nitrogen atmosphere, similar to the thermal behavior observed in poly(acrylonitrile). The vulnerability of the polymer to thermal degradation is caused by the attachment of *HCl* molecules to each amine group. [25]

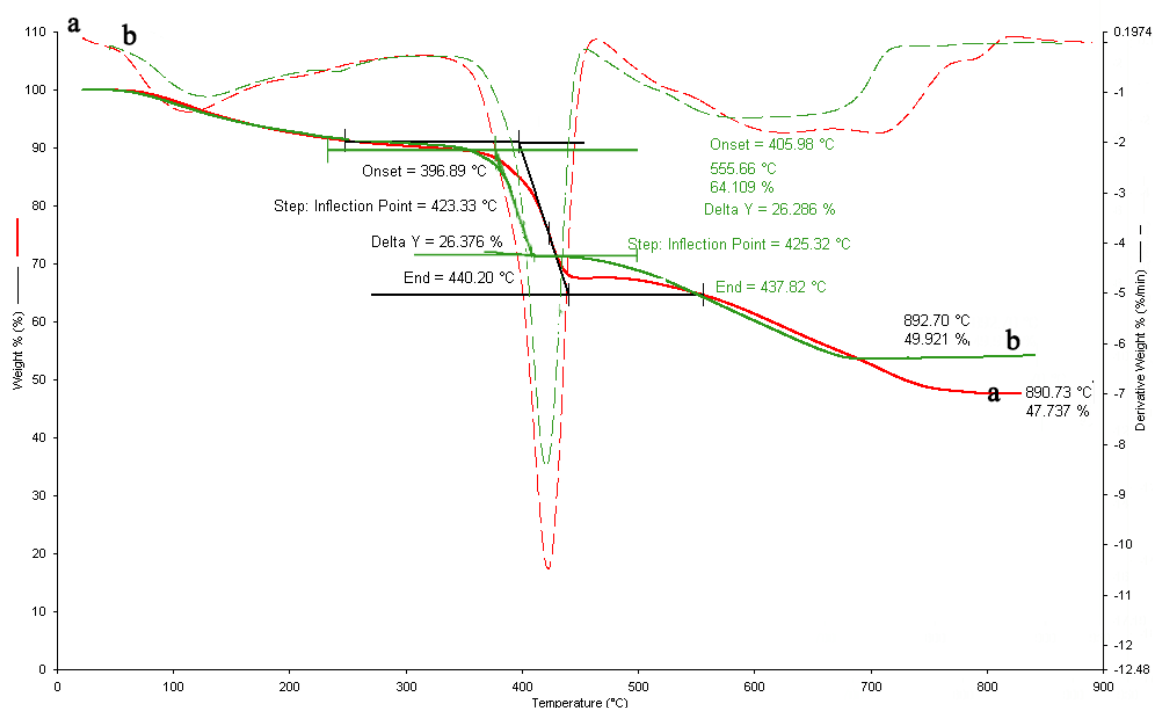


Figure 13. Thermal degradation of **a) *P(VSA)***, **b) *P(VSA.Na)***

The degradation of *P(VSA)* through thermal means follows a 4-step process, with the highest temperature of degradation observed at 420 °C. The initial step, occurring between 90 and 200 °C, is likely attributed to the presence of retained water. The actual degradation takes place in the second step, commencing at 300°C and extending until 450°C. During this stage, approximately 30% of the polymer undergoes degradation. Moving on to the third stage spans from 450 to 650°C. At 600°C, the sample remains non-volatile, while a 5% mass loss is observed at 200°C. When subjected to acid degradation, a significant amount of foam is generated. The fourth and final step initiates at 670°C and concludes at 780°C, with 50% of the

polymer undergoing degradation. Remarkably, even at 900 °C, 47% of the polymer remains intact, as depicted in Figure 13a. [26]

The degradation of $P(VSA.Na)$ can be observed through three distinct processes. The initial process takes place within the temperature range of 100 to 300 °C, resulting in a degradation of 10% of the polymer. The most significant degradation of the polymer chains occurs during the second process, which spans from 380 to 500°C, leading to a degradation of 40% of the polymer. The final step of the degradation process commences at temperatures ranging from 500 to 750°C, leaving behind 50% of the polymer. When the degradation occurs in a nitrogen environment, the foam exhibits the formation of a mineral residue and volumetric expansion at 350°C, as evidenced by TGA analysis (Fig 13b). [26] Furthermore, the thermal properties of the $P(AlAm.HCl)$ and $P(VSA)$ and $P(AlAm.HCl)+P(VSA)$ IPC were investigated.

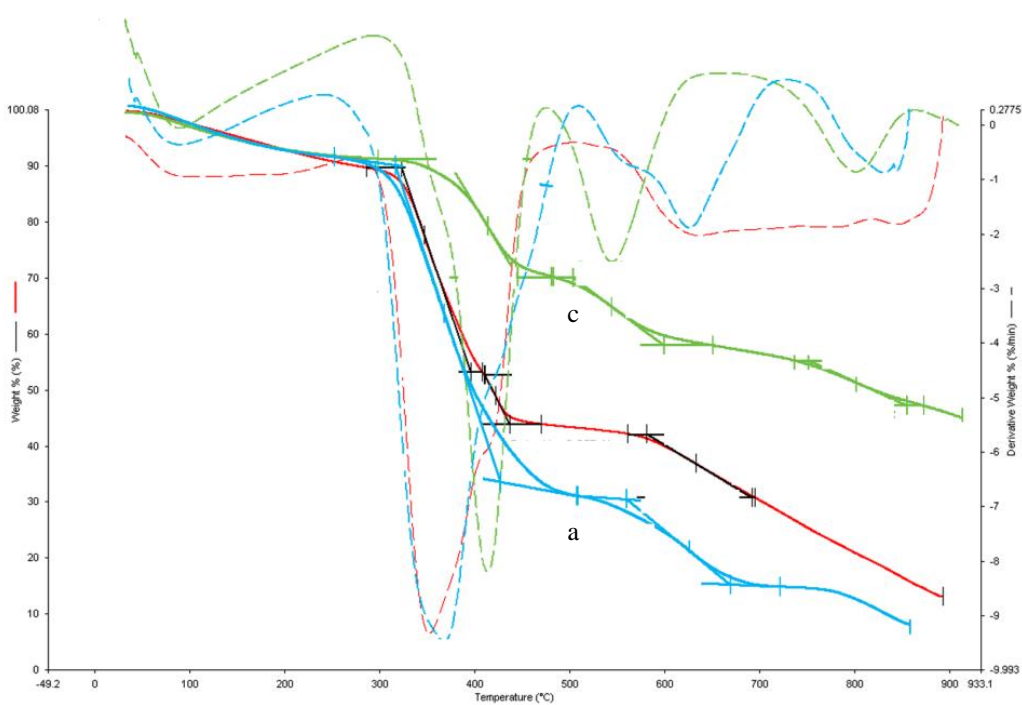


Figure 14. Thermal degradation of $P(AlAm.HCl)+P(VSA.Na)$ IPCs prepared from **a**) (1:1), **b**) (1:0.25), **c**) (0.25:1) mole ratio of feed composition

Figure 14a illustrates the thermal degradation of $P(AlAm.HCl)+P(VSA.Na)$ in a 1:1 mole ratio of feed composition. The degradation process occurs in three steps, with the maximum degradation temperature observed at 347°C. The first step begins at 323°C and continues until 396°C, resulting in a 50% weight loss. Specifically, the degradation of $P(AlAm.HCl)$ takes place at around 300°C, as shown in Figure 15. The second step occurs between 411°C and 436°C, involving the degradation of $P(VSA.Na)$. Finally, the third step spans from 580°C to 894°C. [25-27]

In Figure 14b, the thermal degradation of $P(AlAm.HCl)+P(VSA.Na)$ IPC in a (1:0.25) mole ratio of feed composition is presented. This degradation process follows a two-step phase, with the maximum degradation temperature observed at 380°C. It initiates at approximately 327°C and concludes at around 443°C, resulting in a weight loss of approximately 60%. During this step, the degradation of $P(AlAm.HCl)$ occurs. The second process commences at 582°C and terminates at approximately 694°C, involving the degradation of $P(VSA.Na)$. [25-27]

Figure 14c displays the thermal degradation of the (0.25:1) IPC. This degradation process follows a three-step pattern, with the highest degradation temperature observed at approximately 400°C. It commences around 360°C and concludes at approximately 432°C. Both $P(AlAm.HCl)$ and $P(VSA.Na)$ undergo degradation during this step. The second step begins at around 489°C and finishes at approximately 584°C. Finally, the third step occurs at 732°C and terminates at around 835°C. [25-27]

When comparing the thermal degradation of five IPC polymers with different mole ratios of monomers inter-polymer complex, the (0.25:1) IPC, $P(AlAm.HCl)+P(VSA.Na)$ in different mole ratios of feed composition, emerges as the most significant. This is confirmed by XRD patterns, which can be further illustrated in Tables 1-5.

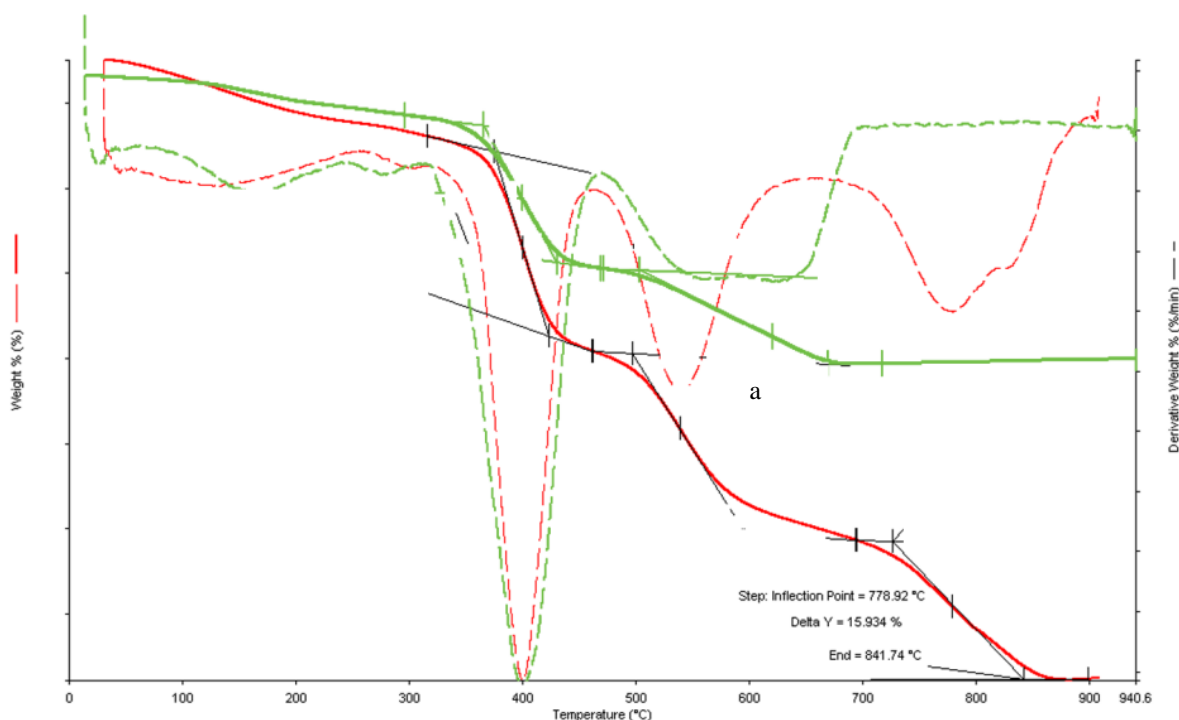


Figure 15. Thermal degradation of **a)** $P(AlAm)+P(VSA.Na)$ IPC prepared from (1:1) mole ratio of feed composition, **b)** $P(AlAm)+P(VSA)$ IPC prepared from (1:1) mole ratio of feed composition

In Figure 15a, the degradation of *IPCs* due to thermal factors occurs through a three-step process, reaching its peak degradation temperature at approximately 400°C. The initial step of degradation takes place within the temperature range of 374°C to 423°C. Subsequently, the second step commences between 496°C and approximately 595°C. Finally, the third step initiates at temperatures ranging from 726°C to 841°C. [25-27]

In contrast, Figure 15b illustrates a distinct thermal degradation pattern for *IPCs*, characterized by a two-step process. The highest temperature at which degradation occurs is measured at 398°C. The initial step of degradation takes place within the temperature range of 362°C to 424°C, while the subsequent step commences between 493°C and 649°C. Figure 15a is an outline to compare the two *IPCs* that are shown in Fig (14a,14b), which show sodium presence at 900°C leads to nearly 53.6% of substance remaining. With sodium in Fig.14a, the pH of the blend *IPC* is changed, and the strength of the hydrogen bonds decreases. However, in Fig 14b, a strong hydrogen bond is formed between two polymers and the initial degradation temperature increases to 374°C. At 900°C, all of the *IPC* polymers are degraded. Tables 6-7 denote the above-mentioned data. [25-27]

Table 3. Data of TGA for IPC in (1:1) P(AlAm.HCl), P(VSA.Na)

	Onset	Inflection point	End
Step 1	321°C	344°C	396°C
Step 2	411°C	421°C	436°C
Step 3	580°C	633°C	892°C 900°C (% 13.12 remained)

Table 4. Data of TGA for IPC in (1:0.5) P(AlAm.HCl), P(VSA.Na)

	Onset	Inflection point	End
Step 1	323°C	374°C	416°C
Step 2	422°C	456°C	487°C
Step 3			900°C (% 15.22 remained)

Table 5. Data of TGA for IPC in (1:0.25) P(AlAm.HCl), P(VSA.Na)

	Onset	Inflection point	End
Step 1	317°C	381°C	443°C
Step 2	580°C	551°C	697°C
Step 3			900°C (% 3.09 remained)

Table 6. Data of TGA for IPC in (0.5:1) P(AlAm.HCl), P(VSA.Na)

	Onset	Inflection point	End
Step 1	327°C	353°C	370°C
Step 2	374°C	407°C	446°C
Step 3	485°C	519°C	575.5 900°C (% 16.3 remained)

Table 7. Data of TGA for IPC in (0.25:1) P(AlAm.HCl), P(VSA.Na)

	Onset	Inflection point	End
Step 1	360°C	401°C	433°C
Step 2	489.5°C	530°C	584°C
Step 3	733°C	782°C	834°C 900°C (% 33.5 remained)

Table 8. Data of TGA for IPC in (1:1) P(AlAm), P(VSA)

	Onset	Inflection point	End
Step 1	374°C	400°C	423°C
Step 2	496°C	538°C	595°C
Step 3	726°C	779°C	841°C 900 °C(%0 remained)

Table 9. Data of TGA for IPC in (1:1) P(AlAm), P(VSA.Na)

	Onset	Inflection point	End
Step 1	362°C	395°C	424°C
Step 2	493°C	604°C	649°C
Step 3			900°C(%53.6 remained)

Conclusion:

Producing layer-by-layer films in the industry, modification of DNA in biology, and preparation of Nano gel and Microgels in drug delivery systems are among the most essential applications of IPCs. We used *P(AlAm.HCl)* and *P(VSA.Na)*, cationic and anionic polymers, to achieve IPCs' most potent complexation properties. We synthesized the polymers using the free radical polymerization method in previous studies. A mixture was prepared with different mole ratios of feed composition to reach the powerful complex. XRD and TGA investigated the prepared IPC to assess the complex power. Strong IPCs were formed in the mole ratio (0.25:1) in feed compositions confirmed by XRD and TGA. FT-IR Spectroscopy was used for the investigation of IPC's structural behavior. The XRD patterns revealed that the IPCs exhibited the highest level of crystallinity at a concentration of 50%. In addition, *P(AlAm)* and *P(VSA)* interpenetrating polymer complexes (IPCs) were synthesized through the removal of HCl from *P(AlAm.HCl)* and the conversion of *P(VSA.Na)* to *P(VSA)*. Both polymers exhibited solubility in water. The presence of NH₂ stretching at 3100 cm⁻¹ was observed in mole ratios of 1:1, while sulfamidization was observed at 1700 cm⁻¹ in mole ratios of 0.25:1 and 1:0.25. The FT-IR spectra obtained from samples prepared at various mole ratios of feed composition and kept at 170°C for 180 minutes (similar to 30 minutes and 60 minutes) showed the presence of the

sulfonamide group, which can be attributed to the influence of temperature. *NMR* spectra confirmed the existence of a weak complexation in the different mole ratios of feed composition. *P(AlAm)* and *P(VSA.Na)* *IPCs* were synthesized based on different mole ratios on a repeating unit basis and were easily dissolved. The structural analysis was conducted using *NMR* spectroscopy. The *XRD* patterns of the *P(VSA)* *IPCs* were formed by the acidless form of *P(AlAm)*. Weak complexation and amorphous structure were found between *P(VSA)* and *P(AlAm)*. Microgel and Nano gel production to apply in the pharmaceutical industry and drug delivery systems is a future research horizon. Using different environmental conditions such as temperature, *pH*, and ionic effect to prepare potent *IPCs* is the other scope of our studies.

Conflict of Interest:

There is nothing to declare.

Financial funding:

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